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### Recommended Citation

Dethe, M. R., Prabakaran, A., Ahmed, H., Agrawal, M., Roy, U., & Alexander, A. (2022). PCL-PEG copolymer based injectable thermosensitive hydrogels. *Journal of Controlled Release*, 343, 217-236. <https://doi.org/10.1016/j.jconrel.2022.01.035>

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Published in final edited form as:

J Control Release. 2022 March ; 343: 217–236. doi:10.1016/j.jconrel.2022.01.035.

## PCL-PEG copolymer based injectable thermosensitive hydrogels

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### Abstract

A number of stimuli-responsive-based hydrogels has been widely explored in biomedical applications in the last few decades because of their excellent biodegradability and biocompatibility. The development of synthetic chemistry and materials science leads to the emergence of *in situ* stimuli-responsive hydrogels. In this regard, several synthetic and natural polymers have been synthesized and utilized to prepare temperature-sensitive *in situ* forming hydrogels. This could be best used *via* injections as temperature stimulus could trigger *in situ* hydrogels gelation and swelling behaviors. There are many smart polymers available for the formulation of the *in situ* based thermoresponsive injectable hydrogel. Among these, poly ( $\epsilon$ -caprolactone) (PCL) polymer has been recognized and approved by the FDA for numerous biomedical applications. More specifically, the PCL is coupled with polyethylene glycol (PEG) to obtain amphiphilic thermosensitive “smart” copolymers (PCL-PEG), to form rapid and reversible physical gelation behavior. However, the chemical structure of the copolymer is a critical aspect in determining water solubility, thermo-gelation behavior, drug release rate, degradation rate, and the possibility to deliver a diverse range of drugs. In this review, we have highlighted the typical PCL-PEG-based thermosensitive injectable hydrogels progress in the last decade for tissue engineering and localized drug delivery applications to treat various diseases. Additionally, the impact of molecular weight of PCL-PEG upon gelling behavior has also been critically highlighted for optimum hydrogels properties for potential pharmaceutical and biomedical applications.

### Keywords

Injectable hydrogel; Thermoresponsive; PCL/PEG copolymer; Tissue engineering; Scaffold

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Declaration of Competing Interest  
No conflict of interest.

# 1. Introduction

“Smart” materials, *i.e.*, materials that can respond to environmental stimuli have recently emerged showing immense potentialities in the biomedical fields. Among them, smart polymeric materials have gained considerable attention in the field of biomedical engineering as a result of their ease of use when compared with smart metals or ceramics [1]. Polymeric smart materials can respond to different external stimuli based on their chemical structure and composition, such as ionic strength, temperature, pH, light, electrical, magnetic, chemical and biological stimulations [2,3]. Therefore, these polymeric materials can be exploited in the development of sensors, drug carrier systems and as well as scaffolds in the field of tissue engineering [4–6]. As a result of the temperature difference outside and within the human body (from ambient to physiological), polymer hydrogels that are triggered by temperature changes have been widely studied [7–9].

In detail, hydrogels are a form of polymeric materials with the ability to imbibe and swell in aqueous solutions. They form three-dimensional polymeric networks with the capability of retaining a large volume of water in their swollen state. The extremely hydrated nature of hydrogels resembling the extracellular matrix characteristics makes them biomimetic materials for tissue engineering [10]. Thermosensitive hydrogels are particularly interesting because a simple temperature change can easily trigger gelation and swelling behaviors. These systems can also be exploited for *in situ* hydrogel formation, for example as drug delivery injectable carriers for local delivery in a minimally invasive manner [11–13]. The drug delivery with PCL-PEG injectable thermoresponsive hydrogel system has many advantages such as high miscibility of the drug in aqueous form due to its low viscosity below body temperature, localized delivery *via* injection enables targeted and time-sensitive delivery of the drug, PCL-PEG makes easier to weigh and disperse due to its powdery crystalline form as compared to the paste-like polyphosphazene and PEG-PLGA [14]. This powder crystalline form results in improved technical convenience.

The drug-loaded PCL-PEG hydrogels also have their limitations as it needs to be formulated above the LCST. Also, homogeneity and poor drug loading into the PEG-PCL hydrogel remain a challenge for scientists. This process is also not suitable for thermolabile drugs. Along with this PCL possesses high crystallinity and hydrophobicity which could hinder the drug release from the hydrogel matrix [15].

To overcome such limitations many researchers have explored the different dimensions associated with the formulations of stable thermoresponsive hydrogel network systems. In general, based on the polymeric source applied for the development of a stable hydrogel network, injectable hydrogels are classified into natural, hybrid or synthetic. To better adjust the properties of hydrogels, they can be cross-linked chemically (*i.e.*, covalently) or physically (*i.e.*, by non-covalent interactions) [16–18]. Various natural and synthetic thermosensitive hydrogels have been extensively investigated, such as chitosan-based [19–21] and polyester copolymer-based hydrogels [22,23], respectively. Polyester-based thermosensitive hydrogels, based on PGA,<sup>1</sup> PLA,<sup>2</sup> PVL,<sup>3</sup> PLGA,<sup>4</sup> and PCL<sup>5</sup> typically

<sup>1</sup>polyglycolide

arranged by joining polyester chain terminals with the ends of polyethylene glycol (PEG) molecules [24].

Among them, PCL<sup>6</sup> is the most preferred one which is also approved by the FDA because of its excellent biodegradability and biocompatibility making it more suitable for application in health care [25]. Generally, it is used alone or in combination with other biocompatible polymers to prepare scaffolds for tissue engineering [26], such as guided bone regeneration membranes [27], carriers for drug delivery [28], and sutures [29]. When PCL, coupled with the hydrophilic PEG, it is possible to obtain amphiphilic thermosensitive “smart” copolymers (PCL-PEG), able to undergo rapid and reversible physical gelation by changing the temperature. By proper selection of block molecular weight and relative amount, PCL-PEG hydrogels may undergo sol-to-gel transition with the rise in temperature ranging from ambient temperature to 37 °C (Fig. 1) [30,31]. The chemical structure of the copolymer is a critical aspect in determining, water solubility, thermo-gelation behavior, drug release rate, degradation rate, and the possibility to deliver a diverse range of drugs, including proteins and bioactive molecules [32,33]. (See Table 1.)

Like PCL, PEGs are also well recognized and approved by FDA to be outstanding biodegradable and biocompatible materials for numerous biomedical applications [34–36]. Recently, thermoresponsive PCL-PEG hydrogels have attracted extensive attention as a promising alternative to the commonly investigated hydrogels based on PLGA-PEG copolymers. However, PLGA-PEG block copolymers suffer from some drawbacks, such as the formation of acidic degradation products upon their rapid degradation, sticky appearance at room temperature making handling difficult, longer dissolution time in the water, and weak hydrogel mechanical properties [37–39]. Due to their crystallinity, PCL-based copolymers have a stable morphology with a significant improvement in handling and transfer [40,41]. Compare to PLGA-PEG hydrogels, PCL-PEG hydrogels have superior rheological properties, slower degradation rate and more extended *in vivo* stability, hence reducing inflammatory reaction and improving the stability of incorporated bioactive molecules [40,42,43]. *In situ* hydrogels are prepared from physical and chemical crosslinking for effective localized drug delivery. Physically crosslinked hydrogels are formed from hydrogen bonding between water-soluble polymers in response to environmental stimuli [44]. On the other hand, chemically crosslinked hydrogels are prepared from Michael’s addition reaction, photopolymerization, and enzymatic reactions [45–48]. These hydrogels are prepared from FDA-approved polymers such as PCL/PEG for biomedical applications [49]. The PCL-PEG copolymer and drugs are dissolved above the melting point of PCL and allowed to form a gel. Hence as mentioned above the PCL-PEG hydrogels are not suitable for thermosensitive drugs for localized drug delivery applications [15]. Moreover, such hydrogels may allow a diffusion-controlled drug release. Thus, thermosensitive hydrogels based on PCL-PEG block copolymers have exciting applications in the biomedical field like scaffolds meant for soft and hard tissue engineering and delivery system for drugs and

<sup>2</sup> polylactide  
<sup>3</sup> poly(δ-valerolactone)  
<sup>4</sup> poly(lactide-co-glycolide)  
<sup>5</sup> poly(ε-caprolactone)  
<sup>6</sup> poly(ε-caprolactone)

proteins too. In this review article, we have emphasized the various developments in the area of PCL/PEG copolymer-based thermoresponsive injectable hydrogels.

## 2. Historical perspectives in the synthesis routes

The slow degradation of PCL polymer, due to its higher degree of hydrophobicity and crystallinity, restricts its clinical application. PCL copolymerization can accelerate the material degradation rate, preserving biodegradability, biocompatibility, and excellent mechanical properties [50–53]. The hydrogels based on amphiphilic PCL-PEG copolymers have numerous biomedical applications, as explained in the following paragraph. The first scientists who prepared the PCL/PEG block copolymers series were Perret and Skoulios [54]. They practically proved the anionic polymerization in the presence of naphthalene-sodium complex as a catalyst for the synthesis. Later, Cerrai et al. introduced a new polymerization method for the synthesis of PCL/PEG copolymer in the absence of a catalyst [55]. In the PCL-PEG copolymers, PCL is the hydrophobic block (A-block) while PEG is the hydrophilic block (B-block). Based on their structure, PCL-PEG copolymers, designated as AB di-block, ABA or BAB tri-block, and star-shaped block structures. For the synthesis of PCL-PEG di-block copolymer, the most recommended method is ring-opening copolymerization from monomethoxy-PEG (MPEG) and  $\epsilon$ -CL in the presence of a catalyst [56–60]. Also, PCL-PEG triblock copolymers can be categorized into two forms: the ABA type which comprises PCEC<sup>7</sup> copolymers and the BAB type comprising PECE<sup>8</sup> copolymers. By the application of ring-opening polymerization, PCEC based copolymers are synthesized from  $\epsilon$ -CL monomers using a dihydroxy PEG as an initiator [61–63]. On the other hand, using coupling reaction, PECE copolymers are synthesized from MPEG–PCL di-block copolymers with a coupling agent, which is a diisocyanate, for example, IPDI<sup>9</sup> and HMDI<sup>10</sup> [35]. These multiblock PCL-PEG copolymers can be synthesized by any of the following methods, *e.g.*, by a two-step reaction, involving the initial synthesis of an isocyanate (–NCO) end-capped prepolymer, by reaction of PCL-diol with a coupling agent (*e.g.*, L-lysine methyl ester diisocyanate), followed by reaction with PEG diol [64]. Also, the PCL-PEG *star*-shaped polymers, synthesized by various methods, *e.g.*, synthetic procedure, convergent, and divergent methods, proposed by Kim et al. [62,65,66]. For the synthesis of PCL-PEG copolymers, different reactants and catalysts can be used to produce various types of di-, tri- or multi-block copolymers of PCL-PEG [31,67].

## 3. Critical gelation concentration

In contrast to the gelation mechanism of hydrogel *via* a chemical method, the mild condition, kept for the sol-to-gel transition of temperature. Under these mild conditions (nontoxic), toxic reagents like organic solvents, residual unreacted monomers, catalysts, crosslinkers, *etc.*, are not used for the formulation [68,69]. The chemical composition of the copolymers and the manner of treatment of the hydrous copolymers can bring about a significant change in the sol-gel transition behavior. Gong et al. investigated the sol-gel

<sup>7</sup>PCL-PEG-PCL

<sup>8</sup>PEG-PCL-PEG

<sup>9</sup>isophorone diisocyanate

<sup>10</sup>hexamethylene diisocyanate

transition and the factors that govern the alteration of the transition diagram. It has been seen that with the increase in the hydrophobic PCL block length, the gelation occurred at lower concentrations and lower temperatures. This may be due to the improved hydrophobicity of the copolymer backbone. Hence it can be attributed that the hydrophobic interactions bring about the sol-gel transition in an aqueous PECE system [70]. As the molecular weight increases, the sol-gel-sol curve shifts towards the elevated temperature, despite the same molar ratio of PCL to PEG [71]. Hence PEG block length is a critical parameter that influences the sol-gel-sol transition curve of the PECE triblock copolymer. The thermoresponsive polymers exhibit their sol-to-gel transition above UCGT<sup>11</sup> or LCGT<sup>12</sup> depending upon the molecular weight of the polymer segment, ratio of hydrophobic and hydrophilic blocks [69]. The orientation of the micelles occurs at a particular temperature and concentration above CGT.<sup>13</sup> For instance, the triblock copolymers consisting of PEG and PCL comprising of BAB or PECE and ABA or PCEC tends to form a hydrophobic PCL micelle core and a hydrophilic PEG shell. The ABA-type PCEC copolymers form a micelle core with inter-micellar bridges, which helps in the gelation of such copolymers at lower temperatures compared to the BAB-type copolymers, which form a hydrophilic shell with a regular hydrophobic core [72]. Apart from this, inverse thermoresponsive hydrogels are comprised of LCGT, a more preferred choice over the UGCT for the application in biomedical. In mild conditions, only hydrogels with LCGT behavior allow the loading of cells and drugs at ambient temperature in an aqueous disperse solution [69].

#### 4. Synthesis

This particular section will cover the synthesis approach responsible triblock and multiblock copolymers. These syntheses depend upon the arrangement of the polymer segments of PCL (A-block) and PEG (B-block) into the form of diblock (AB), PCEC (ABA), or PECE (BAB) triblock, star-shaped, multiblock and graft polymers. In the case of the diblocks, most frequently the ring-opening polymerization is applied in the presence of the catalyst SnOct<sub>2</sub><sup>14</sup> from MPEG<sup>15</sup> and  $\epsilon$ -CL<sup>16</sup> [73]. At high temperatures (185 °C), the diblock of PCL and PEG can be synthesized even in the absence of a catalyst and any solvent [74]. However, by the use of a suitable coupling agent (aliphatic and aromatic diisocyanates), PECE copolymers can be synthesized [75]. The lysine diisocyanate, 1,4-butane diisocyanate, and hexamethylene diisocyanate are some of the most common aliphatic diisocyanates [76]. Another method is to use the equimolar quantity of PCL diols and PEG diacids for the condensation reaction at room temperature [77,78]. Similarly, in the presence of a catalyst and a coupling agent (diisocyanate), PEG/PCL multiblock can be prepared by one-step copolymerization of PCL and PEG. Also, a two-step process was developed for the synthesis of PEG/PCL. This method requires the reaction of PCL diol with a coupling agent (diisocyanate) to initially synthesis an isocyanate (–NCO) end-capped prepolymer under nitrogen followed by the addition of PEG, and the reaction is allowed to continue for 24

<sup>11</sup>upper critical gelation temperature

<sup>12</sup>lowercritical gelation temperature

<sup>13</sup>critical gelation temperature

<sup>14</sup>Stannous octoate

<sup>15</sup>monomethoxy-PEG

<sup>16</sup> $\epsilon$ -caprolactone

h [64]. Another method for the two-step process is coupling the hydroxyl groups of PCEC molecules using terephthaloyl chloride [31,79]. However, these generally used organometallic catalysts for the synthesis of copolymers which are not suitable for biomedical applications. Hence the residual metal present as a contaminant form should be removed completely from the final resulting copolymer [80]. Even though stannous octoate  $\text{SnOct}_2$  is an accepted food additive by the American Food and Drug Administration (USFDA), its cytotoxicity is of great concern for the biosafety of the materials synthesized using this catalyst. Discovering novel tin-free catalysts of high biosafety is another challenge for this field [81]. In line with this, the application of metal-free strategies to carry out the ring-opening polymerization reactions has been developed to synthesize polymers with high biosafety. Additionally, the cost of copolymer production can also be lowered by eliminating the process used for the removal of metal impurities [80]. Enzymatic polymerization is another approach reported by Figueiredo et al., where they employed thermophilic carboxylesterase and Lipase B from *Candida antarctica* to synthesize amphiphilic polyesters for biomedical application [82]. Recently, Qindeel et al. reported a new economical and organic solvent free approach for the synthesis of copolymer using reflux assembly without nitrogen environment and used ice cold double distilled water for extraction and purification of the end product [83]. However, scale up of such ecofriendly approaches for synthesis of PCL/PEG amphiphilic block copolymers remains a major challenge.

## 5. Biomedical applications

Smart, intelligent injectable temperature triggered *in situ* hydrogels, investigated in the form of physical and chemical gels for various biomedical applications. Drug delivery and tissue engineering are the more widely explored areas related to PCL-based injectable hydrogels [84]. This section illustrates the recent developments in the field of PCL-PEG-based injectable thermosensitive hydrogels.

### 5.1. Injectable hydrogels for hard tissue therapy

Reconstruction of damaged/ diseases bones by the use of biomaterials has been widely investigated using bone substitutes [85–88]. In a few studies, ABM<sup>17</sup> (bone substitute) is widely used exhibiting low immunogenicity [89–94]. Various animal models have demonstrated the success of ABM application to support osteoinduction [95]. However, the brittle ABM granules may be released from the defective cavity causing severe inflammation, with consequent pain to the patients [96–98]. In the series of linear thermosensitive hydrogels, PECE triblock copolymers were applied for bone regeneration [95,99,100]. In particular, Ni et al. have combined osteoinductive ABM, and PECE (PEG-PCL-PEG,  $M_n = 3150$ ), forming injectable thermosensitive ABM/PECE hydrogel for the regeneration of cranial bone defects (Fig. 2(G–H)).

The ABM/PECE composite hydrogel was in the sol state at 10 °C, formed an opaque gel at 37 °C and returned a freely flowing solution at 10 °C (Fig. 2(A–F)). The X-radiological examination revealed that the ABM/PECE-based injectable hydrogels can successfully

<sup>17</sup>acellular bone matrix



heal the cranial defect in treated rabbits. The degree of grey-scale displayed by the x-ray imaging and the amount of new bone regeneration was more prominent as compared to its left control. Additionally, the micro-computed tomography validated the findings of the X-radiological examination. The treated right side showed excellent recovery at the 20th week as inferences from the micro-CT images (Fig. 2(I–L)). Also, the prepared ABM/PECE composite was biocompatible and biodegradable having minimal host response as examined after the 4th week of the treatment. Along with this, ABM/PECE-based hydrogels offer various other advantages such as greater ease of handling, broader gelation time (body temperature), noncytotoxic with ROS osteoblasts (*in vitro*) and surrounding (*in vivo*) [102].

Additionally, bone tissue engineering has gained tremendous interest in the field of bone regeneration for various orthopedic surgeries [103–110]. In this context, Fu et al. have prepared a thermoresponsive hydrogel utilizing the triblock PECE copolymer. The PECE solution was added with nanohydroxyapatite (n-HA) to develop a thermosensitive PECE/n-HA hydrogel composite. At the same time, the PECE/Collagen/n-HA tricomponent injectable system has been made to improve cellular attachment [111].

The PECE/Collagen/n-HA composite freely flowed at room temperature and was gradually converted into a gel by increasing the temperature. In detail, PECE/Collagen/n-HA (Fig. 3A) composite system behaved like a low viscous solution (with the low elastic modulus ( $G'$ ) and viscous modulus ( $G''$ )) at room temperature. As the temperature tends to increase  $G'$  and  $G''$  increased up to a maximum value at 37 °C (Fig. 3B), exhibiting gel formation. Further, raising the temperature caused a rapid decrease in  $G'$  and  $G''$  (Fig. 3C). Additionally, *in vivo* experiments have been performed to study the degradation of the hydrogel composite post-surgery effect in a muscle pouch for 7 to 14 days. The composite was degraded entirely after 12 weeks post-implantation. Thus, the injectable PECE/Collagen/n-HA hydrogel represents a potential smart composite injectable hydrogel for reconstructive and orthopedic surgery.

Similarly, guided bone regeneration was achieved by fabricating a bilayer composite material with nano calcium phosphate (CaP) incorporated into PCEC block copolymer and covered with an additional PCL layer. This structural difference compared to the pristine PCEC nanocomposite provided an occlusive barrier for the connective tissue, owing to the presence of the PCL layer. This directs the bone tissue regeneration by hydrophilic CaP incorporated in PCEC layer. The addition of the CaP can reinforce the nanocomposite's mechanical strength, which is crucial for load-bearing bone regeneration. Additionally, the prepared nanocomposites, due to the presence of the CaP provided the nucleation sites for the biological minerals and triggered the mineralization. This, in turn, improved the osteoinductive and osteoconductive properties of the membranes. Hence, it can be said that in addition to the PECE nanocomposites, the PCEC nanocomposites, modified with CoP and extra PCL layer, are potential biomaterials for guided bone regeneration [113].

Apart from these, synthetic thermo-responsive biodegradable polymeric systems based on polyethylene glycol-polypropylene glycol block copolymers (PEO-PPO-PEO) and poly (N-isopropylacrylamide) (PNIPA or PNIPAAm) are being used for delivering the pre differentiated bone marrow stem cells. However, these materials lack osteoinductive

and osteoconductive properties; hence, PCEC/PECE-based copolymers are the preferred candidates for hard tissue therapy application[114].

## 5.2. Injectable hydrogels for soft tissue therapy

Hydrogels are used for various biomedical uses directly just after their preparation (with or without cell) or used after the scaffold preparation. This is because of their unique ability to adopt the characteristic property of the natural tissue. A hydrogel scaffold is meant to support the mechanical structure of a repaired tissue by adhering to the 3D hydrogel framework. Among so many advantages of hydrogels scaffolds, there are associated limitations too. These include irregular seeding owing to the deficient spatial, temporal control; the reduced mechanical strength of the hydrogels limits their use for soft and non-load bearing tissues, and the issue of microvasculature leading to a loss of functional and viability of the seeded cell owing to the insufficient transportation of signaling molecule and nutrients. Apart from these limitations, the hydrogel scaffold remains an interesting topic in tissue engineering. Notably, in the case of repairing articular cartilage lesions by the aid of a combined biodegradable scaffold with chondrocytes [115–118]. In such type of work, small clusters of articular chondrocytes were extracted from the individual patient followed by large-scale cultivation and finally seeded to biodegradable porous polymeric scaffolds [119]. These cell-embedded scaffolds were applied to the defective site for regeneration of articular cartilage. These scaffolds facilitate the mass transfer of nutrients and oxygen *via* their porous structure for sufficient cell seedlings. The addition of ammonium bicarbonate salt particles into the scaffolds produces a highly interconnected pore structure for high cell seeding density [120–122]. For example, Park et al. developed a biodegradable flexible hydrogel scaffold by modulating the block composition and ratio to get properties like swelling ratio, biodegradability, flexibility, elasticity and hydrophilicity [123]. In order to encapsulate the chondrocytes in media, a 3-D culture was prepared to comprise a porous hydrogel scaffold to facilitate penetration of the inner state of the scaffold [124–126]. This precludes the adaptation of the chondrocytes once implanted inside the body [127,128]. The author preferred to use a radical cross-linking reaction of PEG-DA<sup>18</sup> and PCL-DA<sup>19</sup> with sodium chloride salt particulates. This salt leaching produces a highly porous structure of the hydrogel. In addition, both *in vitro* and *in vivo* results (cell culture with rabbit chondrocytes) demonstrate that the cells and tissues exhibited different swelling ratios in the scaffold to facilitate tissue regeneration by efficient cell growth. Hence, in their work, they prove that PEG/PCL-based hydrogel scaffolds could be used for soft tissue regeneration.

In another piece of work, Oroojalian et al. reported the use of thermosensitive PNIPAAm based hydrogel for wound healing. PNIPAAm is a thermosensitive polymer with a transition temperature of 37 °C. However, the hydrogel prepared using only PNIPAAm was fragile and did not provide sufficient mechanical strength [129]. Thus, the author synthesized penta-block copolymer (PNIPAAm-PCL-PEG-PCL-PNIPAAm) for the preparation of hydrogel to achieve sufficient mechanical strength. On the other hand, the investigators synthesized a *star*-shaped copolymer [poly(CL—Co—LA)-b-PEG] and compared the wound healing

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<sup>18</sup>Poly ethylene glycol diacrylate

<sup>19</sup>Polycaprolactone diacrylate

potential of both kinds of polymer-based hydrogels. The cell proliferation and fibroblast cultivation were higher in 20% w/w Penta-block copolymer hydrogel than 10% w/w hydrogel. The prepared Penta-block copolymer showed stable depot on Sprague-Dawley rats and degraded over a month. Besides, the star-shaped copolymer was not recommended for skin tissue engineering as it has low molecular weight and poor mechanical strength, whereas Penta-block copolymer at 20% w/w remained a potential scaffold for wound healing and tissue engineering [130].

In another tissue engineering application of the hydrogel system, PCL-PEG diblock has been merged with another PCL to get a PCL-PEG-PCL triblock copolymer for better cellular compatibility. Noormohammadi et al. have developed polyurethane (PU) (0.1 and 0.25%) based nanocomposites with PCL<sub>500</sub>-PEG<sub>x</sub>-PCL<sub>500</sub> for tissue engineering applications. The molecular weight of PEG used in this study was 1000 and 2000 g/mol to prepare triblock with PCL. In addition, cellulose nanowhisker (CNW) was utilized as a crosslinker, and the prepared modified PU (0.25%) with PCL-PEG-PCL showed better cell compatibility in cellular studies than PU (0.1%) and can be used for tissue engineering [131].

Similarly, Wang and their team investigated on BMSCs<sup>20</sup> to survive and to produce good cell retention to suppress infarction expansion with inhibition of left ventricular (LV) remodeling [132]. These BMSCs were encapsulated in an injectable MPEG-PCL-MPEG<sup>21</sup> hydrogel. They refer to this strategy more because the available implantation techniques are not suitable to offer cell custody in culture medium *via* coronary injection or intramyocardial injection. In addition, these old technique applications are limited because of low cell retention and its survival mainly due to the lack of ECM<sup>22</sup> in the infarcted myocardium [133,134].

In contrast, MPEG-PCL-MPEG hydrogel quickly solidifies to support the internal structure to cure impaired cardiac function and at the same time to reduce the cardiac remodeling too, when injected intramyocardially [135]. Significant cell retention and survival (100%) was observed when performed on rabbit for *in vivo* studies and for *in vitro* MTT assay was used to validate cytotoxicity of the hydrogels concerning BMSCs proliferation. As reported earlier in their studies,  $\alpha$ -CD/MPEG-PCL-MPEG hydrogel composite is not accountable for angiogenesis. However, a hydrogel holding BMSCs upsurses the vessel density in the region of the infarct as compared to BMSCs implanted without hydrogel. Thus, it is clear that intramyocardial injection of BMSCs with hydrogel composite could upsurge the cell holding and vessel density nearby the area of infarction, and prevent LV remodeling thereof.

Interestingly, transforming growth factor-  $\beta$ 1 (TGF  $\beta$ 1) has been reported to have a vital role in the process of chondrogenesis. It is associated with chondrocyte growth, along with the processes involved in the differentiation and augmentation of the biomechanical properties of neocartilage [136–138]. In line with this Tengfei et al. (2017) prepared TGF- $\beta$ 1 loaded PCEC based hydrogel to be exploited in cartilage repair for the first time (Fig. 4A). The sol-gel transition temperature was found to be in the range of 20–50 °C (Fig. 4B–C) with

<sup>20</sup>bone marrow-derived stem cells

<sup>21</sup> $\alpha$ -cyclodextrin/poly (ethylene glycol)-b-polycaprolactone-(dodecanedioic acid)-polycaprolactone-poly (ethylene glycol)

<sup>22</sup>extracellular matrix

excellent syringeability and *in vitro* injectability (Fig. 4D). The storage and loss modulus study reveal the viscoelastic properties of the hydrogel as it represents the elasticity part ( $G'$ ) and viscosity of the hydrogel, further it confirms the TGF- $\beta$ 1 loaded hydrogel stayed at gel phase at 37 °C (Fig. 4E). The study claimed 20% degradation within 3 weeks when the hydrogel was given subcutaneously in rat knee. Also, a sustained release of TGF  $\beta$ 1 was seen throughout 120 h without any initial burst release, while 30% of its part remained inside the hydrogel matrix. Additionally, the macroscopic evaluation study proved that the treatment with TGF  $\beta$ 1 loaded hydrogel resulted in the superior deposition of similar peri-native cartilage tissue, indicating its potential in cartilage defect repair as compared to control over an 8 weeks treatment period (Fig. 4F–K) [139]. However, Saghebasl et al., in their comparative evaluation, found the dominancy of PNIPAAm-PCEC-PNIPAAm over PCEC-based thermosensitive hydrogel towards the cartilage repair. They demonstrated the increased survival rate in cells coated and seeded on scaffolds based on PNIPAAm-PCEC-PNIPAAm hydrogel compared to the PCEC gel alone. Additionally, the PNIPAAm-PCEC hydrogel enhanced the expression of Sox-6, collagen-II, and COMP genes after 10 days of *in vitro* study [140].

### 5.3. Injectable hydrogels for cancer therapy

Cancer represents the prominent reasons for mortality and morbidity throughout the world. As per the report of Global Cancer Observatory by WHO, 2020, cancer is the most prevalent disease with the highest mortality rate and considerably increasing cases year by year. In the year 2020, there are 19,292,789 new cases and 9,958,133 mortalities reported throughout the world. These statistics included the number of all kinds of cancer reported in both sexes all around the globe [142,143]. Numerous treatment options are available like radiation, surgery, hormonal therapy, chemotherapy, immunotherapy, and gene therapy [144,145]. Among these, chemotherapy is the most preferred one to exploit using a cytotoxic drug against cancer cells for the treatment of cancer and inhibition of tumor recurrence [146–148]. The localized drug delivery technique is the most effective one among traditional chemotherapies as it reduces the normal tissue toxicity by avoiding the presence of chemotherapeutics in the systemic circulation. Also, at the same time, it also provides localized action with the sustained release of chemotherapeutics [149,150]. To achieve a localized therapeutic response, a hydrogel carrier is the best approach for efficient drug delivery. Intratumor drug delivery through injectable biodegradable hydrogels remains the most attractive approach which provides a sustained and controlled drug delivery directly into the tumor site and at the same time reduces the systemic toxicity. It also minimizes the associated problem of the chemotherapeutic agents, like poor solubility, and reduces the amount of drug to be loaded into the hydrogel [21,84,151–153]. The requirement of elevated temperature for the unstable anticancer drugs is unsuitable for preparing hydrogel formulation [154]. To translate the thermosensitive injectable hydrogel for anticancer drug from bench to bedside a lot of strategies need to be deployed to overcome the formulation factors like sterilization, scale-up, shelf-life, and improve patient compliance [155]. Additionally, many approved and undertrial anticancer drugs are hydrophobic in nature and most of the injectable hydrogels are hydrophilic in nature. This in turn leads to poor drug loading and entrapment in the hydrogel exhibiting insufficient drug release profile thereof [156]. Also, in order to improve the bioavailability many times high dose of such

hydrophobic agents are required to achieve desired effect at the injection site [157,158]. Hence to address the above in this section, we have highlighted the recent findings towards the PCL-PEG based *in situ* injectable hydrogels for the treatment of cancer.

The modified hydrogel system was also explored to improve the efficacy of drugs and to get sustained release. With this objective, a dual drug-loaded thermoresponsive hydrogel was developed by encapsulating one hydrophobic and one hydrophilic drug, paclitaxel (PTX), and doxorubicin (DOX), respectively. For the development of thermoresponsive hydrogel PEG-PCL was modified into poly( $\epsilon$ -caprolactone-co-1,4,8-trioxo[4.6]spiro-9-undecanone)-poly(ethylene glycol)-poly( $\epsilon$ -caprolactone-co-1,4,8-trioxo[4.6]spiro-9-undecanone) (PECT) by ring-opening copolymerization method. The dual drugs PTX and DOX were loaded into the hydrogel system without affecting the thermo-sensitive property. The increase in viscosity of hydrogel was observed at 31 °C and observed maximum in the range of 35–38 °C. The test tube inverting method showed gel formation at 37 °C. The hydrogel system released DOX by initial burst effect while PTX in a sustained manner. The PTX-DOX hydrogel system offers a fast and intense tumor suppression activity due to burst release of DOX which was prolonged to subsequent sustained release of PTX [159]. Similarly, Yu et al. prepared a thermogelling system based on PLGA-PEG-PLGA block copolymer to sustain DOX delivery. The prepared gel was retained at the target site for more than 20 days upon a subcutaneous injection. However, the viscosity of the hydrogel system was significantly increased after the DOX loading, which is a concerning issue for the injectability of the hydrogel[160]. Also, DOX has been incorporated into the poly(N-isopropylacrylamide) (PNIPAAm) star polymer having an  $\beta$ -CD core and adamantyl-terminated poly(ethylene glycol) based hydrogel. PNIPAAm can strengthen the structural stability of the hydrogels when the temperature is raised from 25 °C to 37 °C. The hydrogel was dissolved very slowly and released the pseudo block copolymer in the form of micelles which further sustained the DOX release encapsulated in the hydrophobic core. This resulted in better cellular uptake even for the multidrug-resistant cancer cells[161]. Wu et al. have worked on the development of thermosensitive injectable hydrogel for the loading of the dual drug, cisplatin (DDP) and paclitaxel (PTX) [162]. These drugs were selected as useful chemotherapeutic agents for the treatment of lung cancer. In general, degradable polymers are preferred for the preparation of the NDDS.<sup>23</sup> Also, MPEG-PCL copolymers are extensively used for the development of micelles towards the delivery of an antitumor drug. In such copolymers, PCL is used as a hydrophobic segment and forms a micelle core for hydrophobic drugs. On the other hand, PEG being a hydrophilic segment forms an envelope and remains as an excellent stabilizing interface [163,164]. Thus, MPEG-PCL micelles represent water-based formulations for the encapsulation of hydrophobic drugs. Micelles may improve the therapeutic efficacy by EPR.<sup>24</sup> Micelles, included in a thermosensitive PECE hydrogel. The objective of the work was the administration of the drug *via* intratumoral injection directly to the lung using PECE hydrogels for controlled drug release of drug-loaded micelles. In detail, PECE hydrogel including DDP and MPEG-PCL micelles loaded with PTX (Fig. 5A) prepared. The resulting hydrogel, coded as PDMP.

<sup>23</sup>novel drug delivery systems

<sup>24</sup>enhancing permeability and retention

To validate micelles, a comparative *in vitro* release test of PTX from free PTX solution and MPEG-PCL/PTX micelles (Fig. 5B) through a modified dialysis membrane. Nearly 100% of the free PTX got released within 4 days, indicating no effect on the dialysis membrane. On the other hand, less than 65% of the drug, released from the MPEG-PCL/PTX micelles up to 2 weeks. This release pattern confirmed the sustained release as compared to free PTX. This feature further established the stable structure of polymeric micelles. To study the gelling kinetics of PECE alone and PDMP hydrogel composite, storage modulus ( $G'$ ) and loss modulus (the  $G''$ ) were measured. At low temperatures,  $G'$  and  $G''$  of PDMP were low, showing low viscosity and fluidity of the composite. As the temperature increases, the solution transformed to an elastic-like gel at the crossover point of  $G'$  and  $G''$ . The composite viscosity was highest at a temperature above 40 °C. Thus, the PDMP hydrogel composite exhibited thermosensitivity. During this study, a consistent thermo-sensitive property of PDPM hydrogel composite as compared to blank PECE hydrogel was observed. This gelling behavior of PDPM hydrogel composite suggested its appropriate administration *via* intratumoral injection. Among various available PCL-PEGs based thermoresponsive hydrogel, PECE composites remain the most preferred one. The PECE exhibits excellent biocompatibility, and at the same time, they are biodegradable too. These composite can very well reduce the initial burst of the drug especially, in the case of an anticancer drug to reduce hemolysis (in the case of an alkylating agent) [21,153]. The PECE composite is therefore very well exploited for the treatment of cancer, also because of its good syringeability and injectability to provide local action. In a similar study, Shim et al. prepared sulfamethazine conjugated PCLA-PEG-PCLA based pH and thermoresponsive block copolymer to achieve PTX sustained release upto 1 month without any burst release. However, the properties of the hydrogel could only be conserved upto 5 mg/ml of PTX loading[166].

Fang et al. developed a delivery system to sustain the release of a hydrophobic anticancerous drug, the honokiol (HK) [167]. In the case of various cancer patients, the common occurrences of malignant pleural effusion are noticed in many cases. This becomes very difficult to treat thereof. The first improved the poor solubility of the HK by converting it into a stable nanoparticle (NP) by using the solvent evaporation method, which is readily dispersed in water [168]. The prepared HK-NP concentration was 25 mg/ml with a size range of about 33 nm exhibiting a good zeta potential of -0.385 mv which is close to neutral. After that, the prepared HK-NP was loaded into the PECE hydrogel synthesized by ring-opening copolymerization technique to produce HK-hydrogel. The PCL/PEG molecular weight was 3408, and the ratio of the triblock copolymer was PEG<sub>480</sub>-PCL<sub>2448</sub>-PEG<sub>480</sub>. An effective administration of the HK hydrogel through intrapleural injection would be responsible for the tumor cell death. This reduced the permeability of pleural vascular and prevented the formation of any more blood vessels in pleural tumors [167]. The release of HK from the HK-hydrogel was only 10% up to 72 h would prevent the exposure of the HK with the healthier tissue and enhance its accumulation at the target site. Thus, the HK-hydrogel efficiently successfully decreased the density of microvessels in the pleural tumors. Similarly, Gong et al. have developed the same thermoresponsive composite of PECE triblock copolymer comprising of different ratios of PCL/PEG to deliver HK [169]. They prepared biodegradable HK-loaded PECE micelles by ultrasonication method



and kept the triblock ratio of PCL/PEG as PEG<sub>5000</sub>-PCL<sub>5000</sub>-PEG<sub>5000</sub> in the absence of any organic solvent. HK micelles were found to be small (~59 nm) and neutral (zeta potential was -0.385 mV) making it more competent to sustain the release of the HK by self-assembly. The cellular uptake of the HK micelles exhibited excellent cytotoxicity towards the cancerous cells, and at the same time, the sustained release was also obtained at a low concentration within the first 48 h. This, in turn, lowers the exposure to the healthy tissue and provides a local effect by increasing the accumulation of the drug at the tumor site. Later on, Wang et al. have also recommended the potential application of the PECE triblock copolymer composite hydrogel as a prominent delivery system for the controlled release of 5-fluorouracil (5-FU) [170]. Wang and their team noticed that peritoneal carcinomatosis (PC) of cancers required an efficient delivery system that would not only deliver the drug at the tumor site but can also reduce the systemic toxicity too [171]. In this context, they selected intraperitoneal chemotherapy as a route to deliver 5-FU to cure colorectal cancer [172]. They synthesized a PECE copolymer composite in the ratio of PEG<sub>960</sub>-PCL<sub>2448</sub>-PEG<sub>960</sub>. The *in vitro* drug release suggested that a lower dose (0.5 mg) of 5-FU from PECE hydrogel exhibited an initial burst (26%) of the drug within 1 h and higher drug release (~80%) in 1 day. This drug release behavior is essential for intraperitoneal chemotherapy, *i.e.*, delayed-release with prolonging retention and effective drug permeation to control the drug release. In addition, the 5-Fu-hydrogel remains an injectable solution at 32 °C and transforms sol-gel at 37 °C confirming the controlled release of the drug thereof. Thus, to control the release of 5-FU the biodegradable PECE hydrogel remains the most prominent delivery system. In another instance, Chantal et al., developed a Poloxamer 407 (PEO<sub>98</sub>-PPO<sub>67</sub>-PEO<sub>98</sub>, MW 12.6 kDa) and Poloxamer 188 (PEO<sub>80</sub>-PPO<sub>30</sub>-PEO<sub>80</sub>; MW 8.4 kDa) based thermosensitive hydrogel to incorporate the 5-fluorouracil as neoadjuvant or adjuvant therapy in colorectal cancer. The T<sub>sol-gel</sub> transition was found at 26 °C and drug release behavior from hydrogel demonstrated an initial burst release of 50% within 1 h which further completely released in 9 h. Also, the *in vivo* study in the mice after intra tumoral injection exhibited reduced systemic toxicity and increased the concentration of the drug at the site of action. However, the mice survival was increased in the adjuvant therapy (after tumor excision) when treated with 5-Fluorouracil hydrogel compared with the 5- Fluorouracil free hydrogel. This study reveals the efficiency of both Poloxamer 407 and Poloxamer 188 based thermosensitive hydrogels for cancer treatment [173].

Similarly, Lei et al. expanded the therapeutic efficacy of PECE hydrogel for drug targeting to achieve the local effect with excellent anti-tumor activity especially in the case of malignant diseases.[174] They investigated the recurrence of the tumor growth in the locoregional after tumor removal especially, in the case of breast cancer [175–178]. To confirm the study, they selected PTX as a model drug and further to increase its solubility, it is converted to micelle first. Thereafter, to deliver the PTX micelle in the target site and loaded to PECE hydrogel (PTX-PECE hydrogel) with different block ration of PEG<sub>2000</sub>-PCL<sub>2000</sub>-PEG<sub>2000</sub>. To confirm the *in vitro* anti-tumor activity, the 4 T1 cell line was used for 3 days. Significant cytotoxicity towards the tumor cell was observed in the case of PTX-PECE hydrogel. To investigate the recurrence of breast cancer, they selected the mice model, and only PTX-PECE hydrogel could delay the recurrence time up to an 18th day. The

drug concentration was significantly increased (70 folds higher than the free PTX group) at the local site in the case of PTX-PECE hydrogel. This study confirms the efficiency of PTX-PECE hydrogel towards the prevention of the recurrence of breast cancer after the tumor resection in the case of the mouse model.

Additionally, injectable PCL-PEG hydrogels were developed for intraperitoneal chemotherapy which remains a promising strategy over conventional intravenous chemotherapy [179,180]. Intraperitoneal combinational therapy requires a suitable delivery system, loaded with the drug. Thus, an ideal delivery system must be able to transport both hydrophilic and hydrophobic drugs [181]. In this context, the dual drug delivery system (DDDS) like thermosensitive hydrogels composed of block copolymers are suitable to protect the drug from degradation, prolong the drug release, and above all to maintain their bioactivity. Based on that, Gong et al. developed a blend of biodegradable PCEC copolymers to produce thermosensitive hydrogels. The authors used a hydrophilic (Fluorouracil (Fu)) and hydrophobic PTX drug, loaded into the PCEC hydrogel [182]. Firstly, PTX was encapsulated into the PCEC copolymer solution in the absence of any surfactant to produce PCEC micelles (PTX encapsulated). Likewise, the PCEC solution was added with Fu [183,184]. Then, the prepared PTX micelle was finally encapsulated into the thermoresponsive Fu-hydrogel forming a stable homogeneous solution. A temperature triggered a noteworthy alteration in  $G'$  and  $G''$  in the case of PTX-micelles loaded Fu-hydrogel was detected. The hydrophilic/ hydrophobic molecular structure balance and macromolecular weight of PECE copolymers significantly influenced the phase transitions. *In vivo* studies validated the formation of the hydrogel when injected subcutaneously into KunMing mice. Following subcutaneous injection, the images were taken on different days ranging from day 1 to day 14 and observed a quick formation of a gel with a spherical shape. To underline the effectiveness, the gel was compared with the Pluronic (F127) gel which wholly dissolved within a few hours. Hydrophilic drug almost got released entirely within a week from hydrogel with high initial burst release; in contrast, the hydrophobic drug showed slower release up to 14 days. The driving forces behind this drug release pattern from hydrogel were diffusion-controlled and degradation/ erosion of the hydrogel [185]. Hence, such hydrogels, used for controlled and sustained drug release.

#### 5.4. Injectable hydrogel for intracameral application

To decrease IOP<sup>25</sup> and at the same time maintain the vision, GFS<sup>26</sup> is the most recommended one [186]. The first line prescribed pharmacologic modulation drugs inhibit fibroblast proliferation. This interferes with the wound healing process and thus diminishes the post-operative wound-healing response. Thus, intracameral injection in the anterior chamber facilitates the diffuse distribution of the drug.

Antiadhesive injectable materials can be useful for ocular applications. For instance, to reduce the IOP for proper vision, glaucoma filtration surgery remains the most effective method [186,187]. Also, postoperative adhesion, avoided by improved surgical techniques. However, postoperative marks with scars are the primary cause of failure; hence modulating

<sup>25</sup>intraocular pressure

<sup>26</sup>glaucoma filtration surgery



agents have to be applied. These agents cause premature wound healing generally by hampering fibroblast proliferation; this decreases postoperative wound-healing development thereof. Therefore repeated/frequent conjunctival injections of the drugs are preferred. Also, modulating agents like 5-FU<sup>27</sup> and MMC<sup>28</sup> cause cell death and apoptosis ensuing in the increased application of angiogenesis inhibiting compounds, acting in contradiction of VEGF,<sup>29</sup> commonly known as anti-VEGF compounds [188–194]. Bevacizumab (humanized monoclonal antibody), applied against VEGF-A [195–197]. Its administration into the anterior chamber remains uncertain while subconjunctival injection produces a secondary therapeutic effect and repetitive injections are thus required [198]. Therefore, to overcome this problem, Han et al. have reported a method for the preparation of an injectable PECE thermosensitive hydrogel [199]. Mainly, intracameral injection of bevacizumab (antifibrotic drug) loaded PECE hydrogel is highly recommended in the postoperative scarring management after GFS in the rabbits and has the potential to avoid postoperative adhesions too [200,201].

*In vivo*, animal studies were performed before and after GFS. With particular concern to filtration bleb and anterior chamber, eyes were observed for post-operative 28 days. The changes seen after PECE thermoresponsive hydrogel is shown in Fig. 6. PECE thermoresponsive hydrogel was utterly absorbed within 3 weeks with no corneal edema. Thus, bevacizumab-loaded PECE hydrogels represent a potential sustained drug delivery system for intracameral injection to prevent post-operative adhesions. However, Rauck et al. studied the bevacizumab release from the PEG-poly-(serinolhexamethylene urethane) (ESHU) hydrogel for 9 weeks. The bevacizumab concentration in the eyes was found to be 4 times higher than the bolus injection. Moreover, the claimed superiority and biocompatibility of ESHU hydrogel over PECE or PEG-PLA-PEG-based hydrogels was more in terms of lactic acid formation as degradation products[203].

### 5.5. Injectable hydrogels for controlled drug delivery

*In situ* thermoresponsive injectable hydrogels are extremely useful when controlled, and a sustained drug release profile is required. The modulation of rheological and drug release as well as degradation properties is necessary for proper hydrogel design. The molecular weight and composition of the copolymer can be varied to modulate the rheological properties [204]. Amphiphilicity is a crucial factor that controls the sol-to-gel transition. Thus, there is a need for a balanced molecular weight of hydrophilic and hydrophobic blocks in the copolymers to have injectability.

The family of Poloxamer or Pluronic copolymers, chemically known as PEG-PPG-PEG<sup>30</sup> triblock copolymers, possesses thermo-responsive behavior and is one of the most explored in the developments of novel carrier systems [12,205]. Nevertheless, the reduced hydrophobicity of PPG block results in a high CMC, which is one of the major problems of such copolymers. Additionally, Pluronic hydrogels are fast eroding and non-biodegradable

<sup>27</sup>5-fluorouracil

<sup>28</sup>mitomycin-C

<sup>29</sup>vascular endothelial growth factor

<sup>30</sup>poly(ethylene glycol)-poly(propylene glycol)-poly(ethylene glycol)

that may cause accumulation of plasma cholesterol and triglycerol in the body thereby inducing systemic toxicity [206]. Such limitations persuaded the search for substitution of PPG block in Pluronic copolymer backbone. The PCL substituted *co*-polymer solved the limitations resulting in reduced CMC, inferior molecular weight after degradation and possible elimination of degradation products from the body [34,207]. PCL is an FDA-approved polymer that is biodegradable and biocompatible and possesses excellent permeability. Following this substitution, Gong et al. have prepared a biodegradable, thermosensitive, injectable hydrogel consuming PECE copolymers for controlled drug delivery [208]. The synthesis of PECE triblock copolymer is a multi-step process that initially involves the synthesis of PEG-PCL diblock copolymer using  $\epsilon$ -CL as a precursor compound. The ring-opening copolymerization was the method of choice, and the process was initiated by MPEG using stannous octoate as a catalyst. After that, this PEG-PCL di-block was further used to synthesize PECE triblock biodegradable polymer by using HMDI as a coupling reagent [182]. The PECE copolymer hydrogel showed a concentration-dependent CGC, LCGT, and UCGT in an aqueous solution. The sol-to-gel transition of PECE hydrogels is thus triggered by the accumulation and packing of micelles above CGC at LCGT [207,209]. A further rise in the surrounding temperature, increases the molecular movement of the PCL block, stimulating reverse transition or gel-to-sol transition [210]. A broader gelation region of PECE, observed with the increasing length of PCL block. Higher molecular weight PCL block caused a decrease in the LCGT (from 35 to 31 °C) and CGC, whereas only slightly increased the UCGT of PECE hydrogel. A hydrogel solution, subcutaneously injected into KunMing mice, was converted into a gel within a few minutes and the gel was stable for 13 days. In the biological environment, the gel started degrading with time to release the drug entrapped, and at 14 days, almost entirely dissolved.

On the contrary, the gel prepared from Pluronic (F127) got dissolved in a few seconds only. Such prolonged effect of PECE hydrogel was attributed to reduced degradation time and increased molecular weight with respect to Pluronic F127. The decreased degradation rate of PECE was due to the behavior of polyester or polyester copolymers [211–214]. Accordingly, the PECE-based injectable thermosensitive hydrogel remained used as carriers for sustained and controlled drug delivery.

There are numerous works reported in the field of thermoresponsive injectable hydrogels in recent past years. Some of the investigators have pointed out the difficulty in injectability of the thermoresponsive hydrogel due to the increased temperature of the needle during insertion into the body [206,215,216]. To overcome this problem, Huynh et al. explored the potential capabilities of pH/temperature-sensitive hydrogels towards successful drug delivery. However, there are a few shortcomings too in using these hydrogels like the drug release pattern through hydrogel follows diffusion-controlled behavior failing to prevent an initial burst release of the drug [217,218]. To prepare this dual-functional system of injectable pH/temperature-sensitive hydrogel, poly (b-amino ester) (PAE), used as a pH-sensitive block and after that incorporated with an amphiphilic PEG-PCL-PEG triblock thermosensitive copolymer. The prepared injectable hydrogel (pH/temperature-sensitive) is a result of the formation of Penta-block copolymer  $\text{PAE}_{1258}\text{-PCL}_{1584}\text{-PEG}_{1650}\text{-PCL}_{1584}\text{-PAE}_{1258}$ . At a low pH (6.5) the solubility of the PAE block increases due to ionization, and at the same time, PEG/PCL exhibited hydrophilicity at low temperature. As the pH

and temperature increase (7.4 and 37 °C, respectively) PAE and PCL block tend to increase their hydrophobicity due to the rise in pH and temperature, respectively. This results in the formation of a micelle structure in the presence of the aqueous environment with PAE-PCLs in the core region surrounded by the hydrophilic PEG groups to form a stable gel phase. To study the drug release behavior, insulin added, and notice that the loading of insulin to the hydrogel, the lower critical gel temperature (CGT) the sol-gel transition decreases with increased insulin concentration. In contrast, the upper CGT and critical gel pH (CGpH) remain unchanged. Further, the *in vivo* studies were performed on male Sprague-Dawley (SD) rats, demonstrating a diffusion-controlled release of insulin from the hydrogel for more than 15 days followed by subcutaneous injection.

In another study, Kim et al. have evaluated RGD peptides conjugated thermosensitive mPEG-PCL hydrogel to study adhesion-dependent cellular behavior. This hydrogel-based RGD peptide was prepared using their own patented technology, and the conjugation of RGD to mPEG (750 g/mol)-PCL was achieved using a nucleophilic reaction. The hydrogel formulation retains as a suspension at ambient temperature. Sol-gel phase transition occurred when the temperature increased at 37 °C with a 2.4 mM concentration of RGD peptide-containing hydrogel system. The RGD conjugated peptide significantly enhanced the focal adhesion and thus facilitated the cytoskeleton reorganization through an integrin-mediated cell signaling mechanism. The mPEG-PCL injectable hydrogel provided sustained release [219].

Later on, similarly, Huynh et al. extended this investigation to examine the efficacy of the delivery system on streptozotocin (STZ)-induced diabetic rats [220]. Briefly, STZ was administered to SD rats by intraperitoneal injection to damage insulin secretory cells in the pancreatic islets. The results showed that only 30 wt% of copolymer solution (Penta-block) after subcutaneous injection of a small dose of insulin (10 mg/mL) into STZ-induced diabetic rats successfully maintained the glucose level in blood in a healthy range up to 7 days, with no loss in the body weight.

Utilizing the PCL/PEG blocks ahead for the drug delivery, Khodaverdi et al. synthesized a PCL-PEG-PCL triblock copolymer by both conventional and a rapid microwave-assisted method to produce an injectable solution to control the release of drug from gelling matrix [221]. To validate the delivery system, AMPB<sup>31</sup> and VANH<sup>32</sup> were used as model drugs. For synthesis, the molecular weight of triblock copolymer solution was kept as PCL<sub>1500</sub>-PEG<sub>1500</sub>-PCL<sub>1500</sub> and PCL<sub>1000</sub>-PEG<sub>2000</sub>-PCL<sub>1000</sub> both by a conventional method and microwave-assisted method. Among these two PCL<sub>1500</sub>-PEG<sub>1500</sub>-PCL<sub>1500</sub> exhibited higher sol-gel transition found to be good and selected for further study due to shorter PCL segments and low PCL/PEG ratio. The chosen copolymer solution releases the drugs by bulk erosion more in spite of the diffusion control from the copolymer. The PCL<sub>1500</sub>-PEG<sub>1500</sub>-PCL<sub>1500</sub> hydrogel successfully released the active ingredients for more than 15 days. The nature of the drug however affected the drug release like, VANH being a hydrophilic drug easily bound to the PEG hydrophilic segments which in turn reduced

<sup>31</sup>amphotericin B

<sup>32</sup>vancomycin Hydrochloride

the number of PEG segments and increases the formation of the hydrophobic bond at a lower temperature and *vice versa* is for AMPB. Thus, the PCL<sub>1500</sub>-PEG<sub>1500</sub>-PCL<sub>1500</sub> injectable thermogelling system is more capable of controlling the release of the drug. Extending the remarkable efficiency of PECE thermoresponsive injectable hydrogels, Gong et al. synthesized PECE triblock copolymer in the ratio of PEG<sub>550</sub>-PCL<sub>2000</sub>-PEG<sub>550</sub> by ring-opening copolymerization of  $\epsilon$ -CL [99]. Briefly, the synthesis of PEG-PCL di-block was undergone in the presence of MPEG using catalyst stannous octoate, and this PEG-PCL di-block was used for further synthesis using HMDI as a coupling agent to produce PECE triblock. This PECE hydrogel exhibited a sol-gel transition at 37 °C. To understand the drug release behavior Vitamin B12 (VB12) was used as a model drug. This hydrophilic drug molecule exhibited complete release almost in 7 days with a higher release rate (90% in 24 h) with high initial burst release (20% in 1 h) from PECE hydrogel. This drug release mechanism from the PECE hydrogel is mainly due to the erosion and diffusion effect of the hydrogel [185]. In this series, Gong et al. also worked on successfully validating the gel formation and its degradation by injecting the PECE solution subcutaneously into KunMing mice [208].

Buwalda et al. have explored another dimension not only to increase the mechanical strength but also to improve the stability of the hydrogel systems. The work relates to the extension of the polymeric chain/block to strengthen the entire hydrogel system. Interestingly, an 8-armed star block copolymer hydrogel system is deployed with the initial replacement of the PLA with the PCL group. The stannous octoate catalyzed ring-opening polymerization of the  $\epsilon$ -caprolactone method was used to synthesis 8-armed PEG-PCL star block copolymers linked with amide group between PEG and PCL in toluene at 110 °C. The gel stiffness strongly depends on the hydrophobic block length. Also, the increase in  $G'$  is due to an increase in caprolactone block which remains more densely crosslinked. This is mostly due to the formation of intermolecular H-bonds between amide groups. Normally, T<sub>sol-gel</sub> is increased is due to higher concentration and temperature, between the cross-linked micelles. Physical crosslinking of micelle plays a vital role in gel-sol transformation, as at higher concentrations gel to sol transition is observed with an increase in temperature. *In vitro* stability of PEG-PCL star block copolymer hydrogels was significantly higher than PEG-PLA star copolymers hydrogels due to a lower rate of hydrolytic degradation of PCL than PLA.

Steinman et al. have studied the effect of increasing molecular weight of the polymer (PEG) on gelling properties, rheology, and stability of the hydrogels. For this study, PCL-PEG-PCL triblock copolymers were synthesized by ring-opening polymerization of caprolactone with a different molecular weight of PEG blocks like PEG 2000, 4000, and 8000 Da. It was observed that increasing the molecular weight of PEG blocks results in the formation of stiff gels upon increasing temperature. However, solubility will be compromised upon increasing the molecular weight of the copolymer which will enhance its hydrophobicity. Based on rheological studies the higher molecular weight PEG (i.e 8000 Da) shows stronger  $G'$  (storage modulus) than the low molecular weight PEG (i.e 4000 Da) and indicates the Pseudo plastic type of flow behavior. However, the investigator found that with low molecular weight PEG (*i.e.*, 2000 Da) the hydrogel was not formed and had no gelling effect. The T<sub>sol-gel</sub> transition was found after heating at 25 °C with high molecular weight

PEG (i.e 8000 Da). In all, replacing the hydrophobic PLGA blocks with PCL gives advantages such as increased mechanical strength, biodegradability, and a better safety profile.

Utilizing this strategy, Nguyen and the group developed *in situ* injectable hydrogels for the controlled delivery of insulin. The *in situ* gelations occurred upon injection into the body in response to the endogenous stimuli, pH and temperature, by altering the Penta-block copolymers of PCL-PEG. The synthesis of Penta-block copolymer takes place by conjugation between the hydroxyl group of thermoresponsive triblock copolymers (poly( $\epsilon$ -caprolactone)-b-poly(ethylene glycol)-b-poly( $\epsilon$ -caprolactone)) (PCL<sub>1845</sub>-PEG<sub>2050</sub>-PCL<sub>1845</sub>) and carboxylic group of oligo serine. Owing to the stimuli-responsive properties, the penta-block copolymers form a gel at physiological pH and temperature (7.4 and 37 °C) while remained soluble at higher pH. It was observed that a higher concentration (30% w/w) of the Penta-block copolymer resulted in stable gel which showed sustained degradation. On another side, insulin-loaded chitosan nanoparticles were prepared using the electrospraying method and mixed with the Penta-block copolymers for controlled delivery. The insulin-loaded chitosan nanoparticles with Penta-block copolymers showed release over one month in a controlled manner with inhibition of initial burst release of insulin. The investigators believe that this formulation might also control the release of other proteins with the reduction of early burst release [222]. The PCEC-based hydrogel is more superior in delivering insulin than the PLGA<sub>1000</sub>-PEG<sub>1500</sub>-PLGA<sub>1000</sub> based hydrogel (ReGel®), where its drug release was controlled only for up to 15 days[223].

PCL-PEG hydrogels have been developed in the form of supermolecular hydrogels for drug delivery. The responsible phenomena for supramolecular hydrogel systems are well-organized noncovalent intermolecular binding interactions between two or more linear polymers and CDs.<sup>33</sup> These supramolecular hydrogels are explored as CDDS.<sup>34</sup> Here an example of supermolecular hydrogels for lysozyme delivery is described. Lysozyme has mostly been applied in medicine, food and pharmaceutical industries because of its antibacterial ability against Gram-positive bacteria [224]. It is also responsible for the modulation of the TNF,<sup>35</sup> stimulation of some interleukins and initiating the activity of phagocytizing cells [225]. However, it is also responsible for antiviral activity against several viral strains. The only limitation associated with its use is the complex physical and chemical instabilities, which remains the biggest challenge for its formulation and delivery. Thus, a suitable technique is required which involves no chemical reaction, no organic solvent, and ease in the formulation. In the same context, there is a need to fabricate supramolecular hydrogels [226]. Working on this concept, Ma et al. successfully encapsulated lysozymes to sustain its release from a supramolecular hydrogel [227]. To prepare this supramolecular hydrogel,  $\alpha$ -cyclodextrin ( $\alpha$ -CD) and PCL-*b*-PEG diblock copolymer were prepared in the presence of chicken egg, lysozyme. This hydrogel did not require any crosslinking agent and was formed at room temperature as supramolecular hydrogel (Fig. 7).

<sup>33</sup>cyclodextrins

<sup>34</sup>controllable drug delivery systems

<sup>35</sup>tumor necrosis factor

A crystalline inclusion complex was formed by  $\alpha$ -CD and low molecular weight of PEG or PEO<sup>36</sup> in an aqueous solution. The hydrogel's physical formation was due to the aggregation into microcrystals of PEG blocks in PCL-*b*-PEG copolymer and also to the aggregate of  $\alpha$ -CD inclusion complexes [229–231]. A low amount of PCL-*b*-PEG and  $\alpha$ -CD led to the formation of the low viscous solution, and the solution transforms into a gel after 12 h. On the contrary, in the case of optimized higher concentration of PCL-*b*-PEG and  $\alpha$ -CD, gelation time was rapid (2 min). The formation of the polymeric network in the supramolecular hydrogel controlled the release of the encapsulated lysozyme and produced a stable hydrogel with excellent biocompatibility and excellent mechanical properties. Medical imaging has an essential role in implant research. The implant can be monitored *in vivo* by the use of US,<sup>37</sup> CT,<sup>38</sup> and MRI.<sup>39</sup> Additionally, fluorescence imaging can be applied because of its high sensitivity, low cost, low level of radiation and non-invasiveness.

Lv et al. synthesized a PEG–PCL copolymer-based thermosensitive porphyrin hydrogel [232]. In this system, a fluorescence tag for *in vivo* implant imaging, *i.e.*, porphyrin (POR) was included, and the material was tested in a mice model. POR can be used for the formulation of hydrogels by coupling POR with PEG (Fig. 8). A similar strategy was also explored by Lovell et al. to develop a hydrogel by forming Pd and Cu complexes using PEG diamine and mesotetra(4-carboxyphenyl) porphyrin [233,234]. POR-conjugated PEG started the ring-opening copolymerization of  $\epsilon$ -CL<sup>40</sup> using (Sn (Oct)<sub>2</sub>)<sup>41</sup> catalyst to synthesize POR–PEG<sub>1000</sub>–PCL copolymer (Fig. 8).

The four functional groups present in the porphyrin easily substitute the PEG after copolymerization by PCL. Due to this conjugation of the porphyrin, POR-PEG-PCL hydrogel displayed sol-to-gel transition behavior (at 30 °C) and PCL-PEG-PCL hydrogel present the same (32 °C). This approach could be exploited to regulate final molecular weight and to improve fluorescence intensity. The tube inverting test method was used for the estimation of sol-gel-sol phenomena of POR-PEG-PCL copolymers. An intermediate concentration of copolymer resulted in the quick gel formation of triblock copolymer solution of POR-PEG-PCL within a minute. This was because of the star structure (with high molecular weight) and macrocycle structure of porphyrin. In conclusion, the porphyrin-incorporated thermosensitive hydrogel could form fluorescent injectable materials. Such hydrogels are extremely useful for preclinical testing allowing monitoring of hydrogel degradation and hydrogel molecule migrations to different organs, after injection.

Injectable copolymers were also formed by coupling Pluronic polymers and PCL, obtaining PCL-Pluronic-PCL copolymers [236,237]. In detail, Liu et al. successfully synthesized a series of the PCL-Pluronic-PCL biodegradable block copolymers. These triblock copolymers are synthesized by ring-opening polymerization of  $\epsilon$ -CL initiated by

<sup>36</sup>poly (ethylene oxide)

<sup>37</sup>Ultrasonography

<sup>38</sup>tomography

<sup>39</sup>magnetic resonance imaging

<sup>40</sup> $\epsilon$ -caprolactone

<sup>41</sup>stannous octoate



a macromer PEG-PPG-PEG (Pluronic) (Fig. 9) [238]. The formulated copolymers showed a temperature-dependent sol-to-gel transition in water.

By increasing the molecular weight of PCL blocks, the LCTST<sup>42</sup> increased because of an increase in the melting point of associated copolymers. However, the UCTST<sup>43</sup> decreases due to an increase in hydrophobic interaction among the copolymers. The sol-to-gel transformation was based on the thermogelation mechanism like partial crystallization, micelle packing, coil-to-helix transition, hydrophobic association [239,240]. The author proposed a double layer shell-core structure of copolymer micelle responsible for the sol-to-gel transition of PCL-Pluronic-PCL copolymer (Fig. 10). Accordingly, strong hydrophobic PCL blocks due to hydrophobic interaction constituted the core, on the other hand, hydrated PEG block and weak hydrophobic PPG block formed the inner shell and the outer shell, respectively. The entrapped water is present in the entangled PEG block, and little water is present in the PEG block [241] and the untwisted PPG blocks [242,243]. Thus, in this work, the authors have developed a method to produce a double layer shell-core structure of PCL-Pluronic (L35)-PCL micelles, explaining the possible mechanism responsible for sol-to-gel transition, and proposed its use as an aqueous injectable system designed for a controlled drug carrier. Additionally, Jun Li et al. prepared the supramolecular hydrogel system, suitable for the sustained release of drugs. The hydrogel was self-assembled between a triblock copolymer of PEO-poly[(*R*)-3-hydroxybutyrate]-PEO. (PEO-PHB-PEO) and  $\alpha$ -cyclodextrin. Two triblock copolymers were prepared with different molecular weights such as PEO-PHB-PEO (5000–3140-5000) and PEO-PHP-PEO (5000–17,500-5000) and resulted in gel formation at the body temperature. Dextran was used as the model drug. The controlled release dextran FITC was found with the high concentration of  $\alpha$ -CD (9.7 wt%) of  $\alpha$ -CD-PEO-PHB-PEO (5000–3140-5000) hydrogel for 1 month. The prepared supramolecular hydrogel has the potential to get the sustained/controlled release of high molecular weight drugs with injectable thermosensitive hydrogels [244].

## 5.6. Injectable hydrogels for post-operative treatment

To prevent postoperative intestinal adhesion, the biodegradable and thermoreversible PCLA<sub>1730</sub>-PEG<sub>1500</sub>-PCLA<sub>1730</sub> triblock copolymer hydrogel was prepared by Zhang et al. (2011) (Fig. 11A). Ring-opening polymerization methods were used for the synthesis of PCLA<sub>1730</sub>-PEG<sub>1500</sub>-PCLA<sub>1730</sub> triblock copolymer hydrogels. The hydrophobic blocks were formed by adding  $\epsilon$ -Caprolactone (CL) with D, L-lactide (LA) to reduce the acidic effect of material degradation. The T<sub>sol-gel</sub> transition of the concentrated PCLA<sub>1730</sub>-PEG<sub>1500</sub>-PCLA<sub>1730</sub> aqueous solution takes place with a temperature shift from room to body temperature (Fig. 11B & E).

The hydrogel was formed within 10–20s after injecting the aqueous solution into the rabbit model (Fig. 11C). The *in situ* hydrogel was found stable for about 12 weeks *in vitro* (Fig. 11F) and 6 weeks *in vivo*. In this study, the investigator found that the prepared gel starts degrading after 28 days upon subcutaneous injection into the *in vivo* rabbit model while taking around 56 days in the *in vitro* studies (Fig. 11G). Alongside, it

<sup>42</sup>lower critical transparent sol temperature

<sup>43</sup>upper critical transparent sol temperature

demonstrated minimum hemolytic (Fig. 11D) and cytotoxic effects. The effect of prevention of post-operative adhesion was evaluated in a rabbit model of sidewall defect and bowel abrasion and it was confirmed that after surgery for thirty days, the control group and the commercialized anti-adhesion film fails to the prevention of adhesion as compared with the hydrogel (Fig. 11H–M). This behavior indicates the injectability of the developed tri-block polymer PCLA<sub>1730</sub>-PEG<sub>1500</sub>-PCLA<sub>1730</sub> [204]. Thus, the PCLA<sub>1730</sub>-PEG<sub>1500</sub>-PCLA<sub>1730</sub> thermosensitive hydrogel remains potential for the prevention of post-operative adhesion as it is convenient in operation and more effective in reducing the formation of intraperitoneal post-operative intestinal adhesion.

## 6. Future perspectives and challenges

PCL-based *in situ* injectable thermoresponsive hydrogels are a broad class of biomaterials that share a high degree of biocompatibility and biodegradability and suitability for minimally invasive treatments. Such hydrogels are prepared by a simple method, involving dissolution in water solution followed by gelification through a change in the temperature, with no toxic or side effects during *in vivo* hydrogel formation. In recent years, the scientific community has widely dealt with the development of *in situ* injectable thermoresponsive hydrogels for various biomedical applications, including cancer therapy, drug release, and tissue regeneration. Since the methods of preparation of these hydrogels are simple, they could be widely applied in clinics in the future, if available at low cost. Hence, efforts should be made to minimize the cost of copolymer synthesis. Up to now, mainly pre-clinical investigation has been carried out on such hydrogels with ample evidence of their promising properties. These hydrogels are still in a ray of hope towards success in clinical trials. By this review article, we would like to draw the attention of the scientists working in this area to make available these injectable systems in the market at the lowest price, compared to the conventional drug dosage forms like tablets, capsules, *etc.* These injectable systems for drug delivery could be advantageous for local drug therapies, *e.g.*, for the controlled release of chemotherapeutics after tumor surgery. Furthermore, they can be applied in combination with bioactive molecules for soft and hard tissue regeneration, as injectable scaffolds, or *vice versa* as non-adhesive injectable systems to inhibit tissue adhesion. All of these approaches are advantageous for their minimal invasiveness. Additionally, thermosensitive polymers, including those based on PCL-PEG copolymers, could be explored for new emerging applications such as the additive manufacturing of scaffolds with controlled architecture for tissue regeneration or tissue modeling [258]. Despite these advantages, hydrogels are associated with several limitations like lack of mechanical strength in aqueous media, which further affects its load-bearing capacity. Also, the hydrophilic chain of polymers and excessive water content are thermodynamically incompatible with the hydrophobic moieties and hence these are not suitable for the delivery of hydrophobic drug cargo [259].

## 7. Guideline for the future

In the development of an *in situ* thermoresponsive hydrogel, different polymers have been investigated for various biomedical applications like hard/ soft tissue engineering and drug delivery systems. Among these, PCL copolymer-based thermoresponsive hydrogels remain the most promising due to their excellent biocompatibility and biodegradability. PCL



copolymerization with PEG allows tailoring the hydrophilicity and crystallinity degree of the material, enhancing biocompatibility and increasing the degradation rate while preserving good mechanical properties. PCL-PEG-based thermoresponsive injectable hydrogels have been proven as efficient carrier systems for various biomedical applications because of their ease of formulation, amphiphilic nature, better stability in biological milieu, enhanced cargo loading, and excellent biocompatibility. These hydrogels are very well exploited as a carrier for cells in tissue engineering, self-healing material and a promising carrier system for the delivery of biomolecules and drugs. Hydrogel gel strength should be regulated depending on the tissue to be treated, for example in the case of bone regeneration; injectable composite hydrogels can be used. The type of PCL-PEG copolymer (di-, tri- or multi-block copolymer) can affect the resulting mechanical strength, due to differences in molecular weight and crystallization degree of the copolymers. Additionally, different types of copolymers possess different degradation rates making them suitable for various applications. Hydrogel physicochemical properties can be easily altered by the blending or functionalization with bioactive molecules guiding cell behavior, such as proteins or bioactive peptides. Besides the possibility to engineer injectable scaffolds, PCL-PEG hydrogels can be exploited for the preparation of novel scaffolds with rapid prototyping. Such applications will deserve further investigations in the future.

## Acknowledgment

This present compilation is possible because of the facilities and infrastructural support provided by the National Institute of Pharmaceutical Education and Research, Guwahati, Assam, 781101. One of the author wants to acknowledge funding from National Institute of Health, USA (R15NS108815 and R01AI147731).

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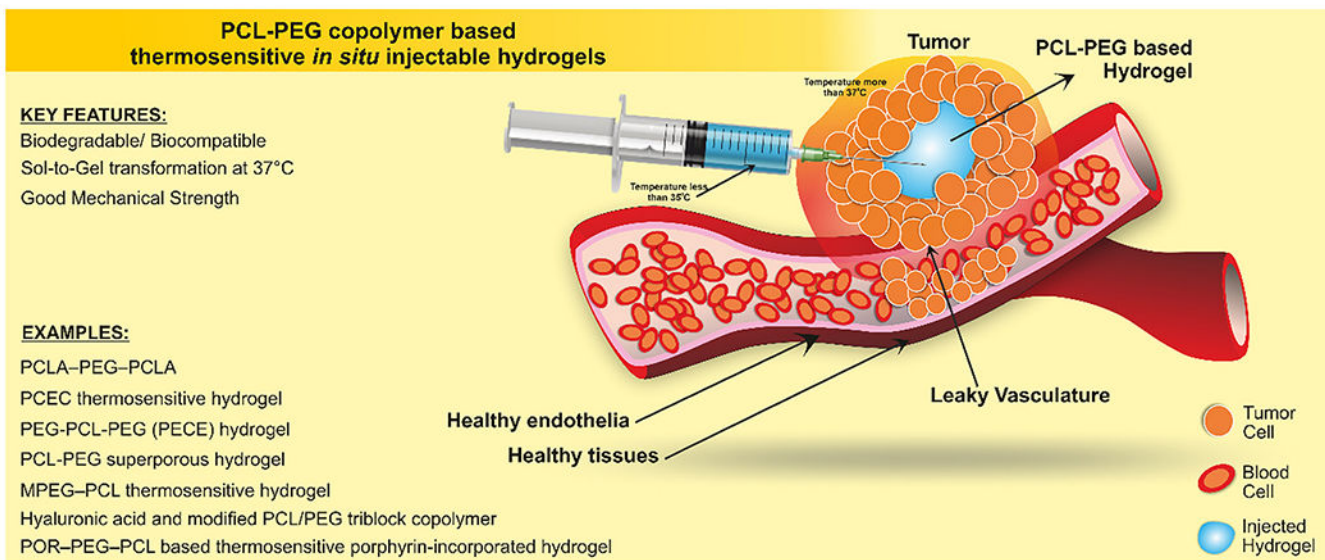


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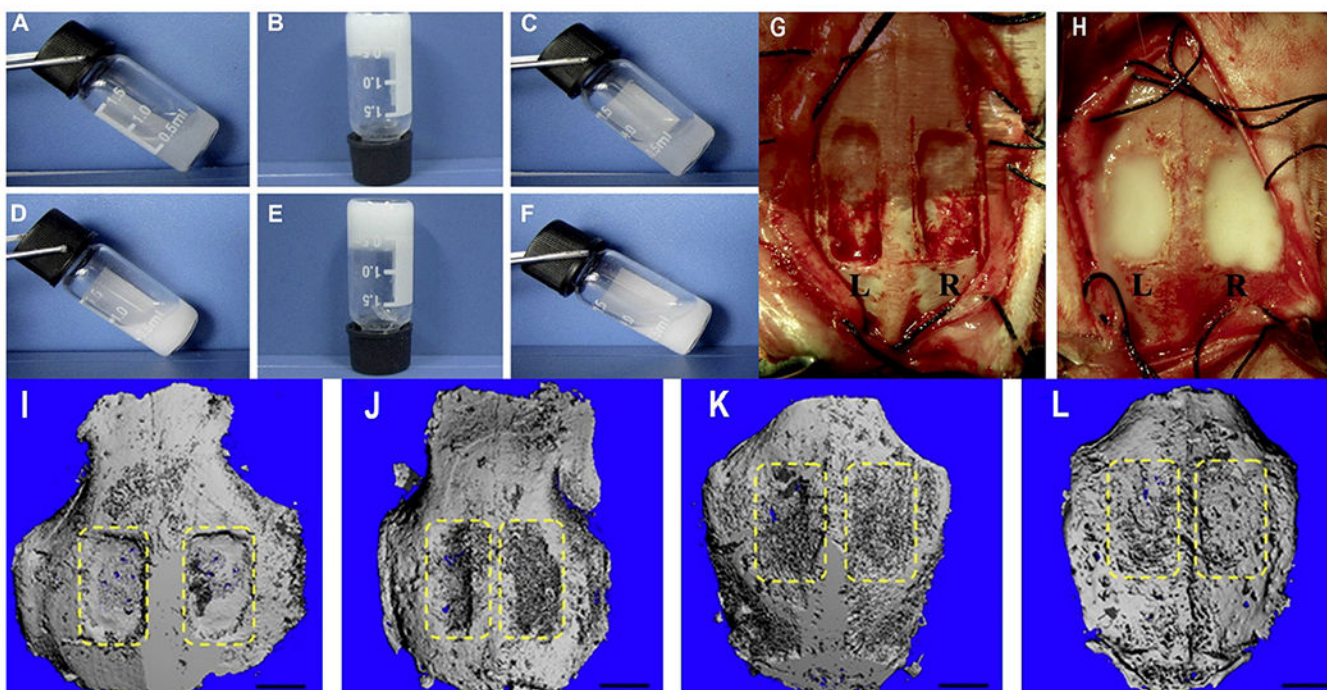


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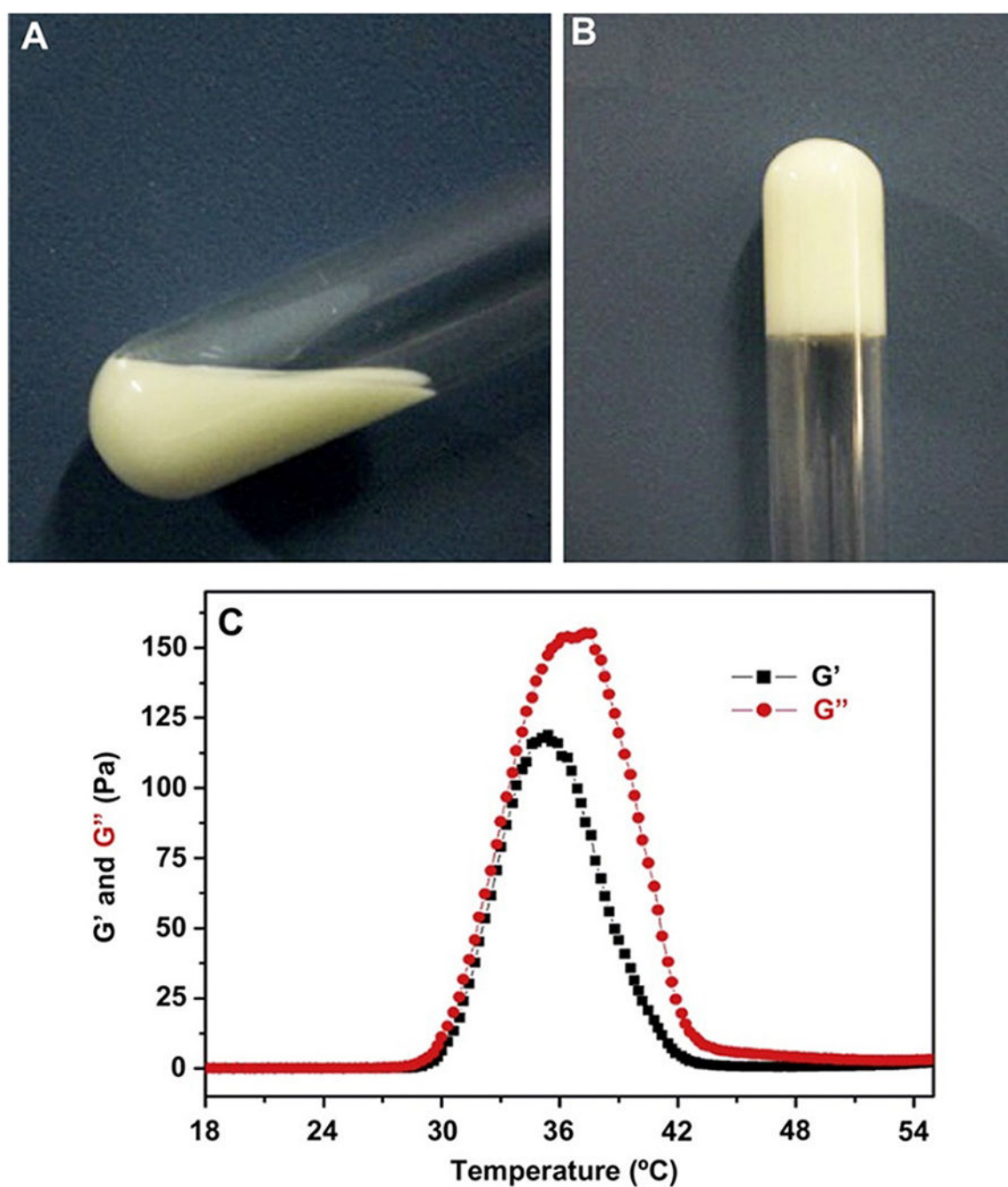


**Fig. 1.**  
 PCL-PEG copolymer based thermosensitive injectable hydrogel. The elevated temperature at the site of the tumor gelled the hydrogel instantly to minimize the initial burst of the drug at the site of delivery.



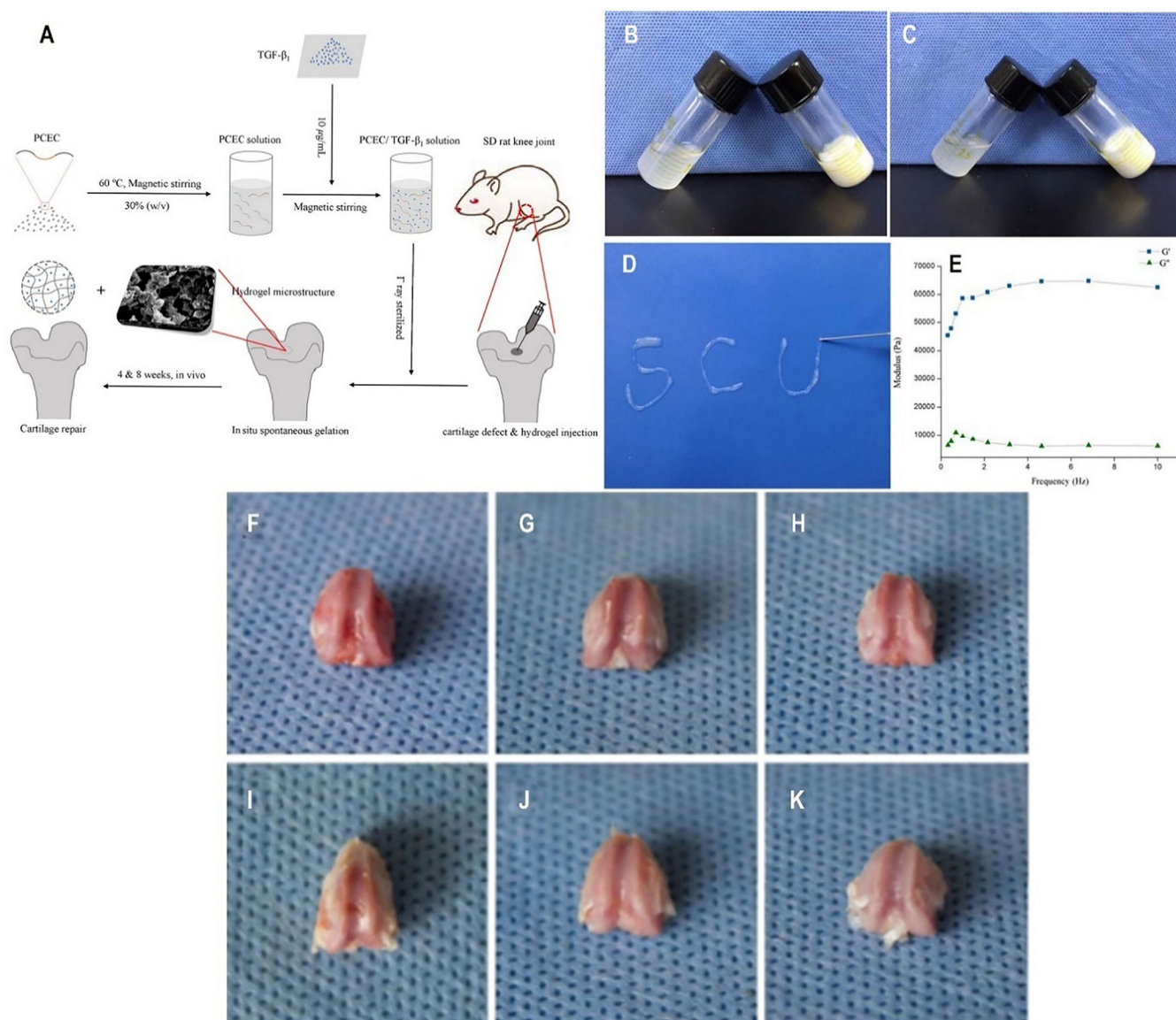
**Fig. 2.**

(A-F): Thermosensitivity of ABM/PECE hydrogels at various temperatures. The composites can be seen in sol state at 10 °C (A&D), opaque at physiological temperature 37 °C (B&E) and return to free-flowing sol at 10 °C (C&F). A-C: Pure PECE; D-F: ABM, 30 wt%. G&H: Surgical procedure. G: Filling of the ¼ rectangular cranial defects (12 mm × 8 mm) in New Zealand White rabbits labeled as L left, pure PECE; R ¼ right, ABM/PECE composite. Both the pure PECE (G) and the ABM/PECE composite (H) were gelled in a few minutes at the body temperature. I-L: 3D Micro-computed tomography of cranial defect shows the healing of the cranial defect at various stages of treatment, (I) Surgery moment, (J) 4 weeks, (K) 12 weeks, (L) 20 weeks. [Adopted and modified with permission from Ni et al. [101]].

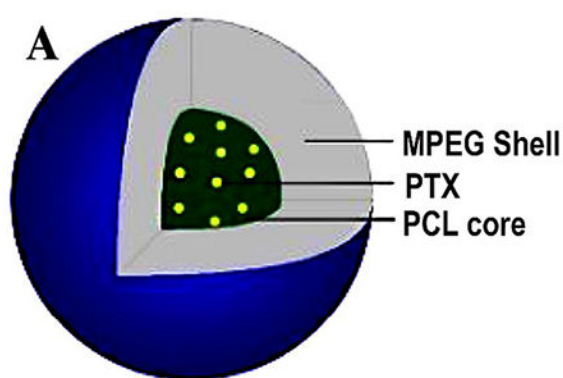


**Fig. 3.** Injectable hydrogel composite is showing thermal sensitivity. Fig. 3 (A) and (B) show gel formation at 20  $^{\circ}\text{C}$  and 37  $^{\circ}\text{C}$  respectively. Fig. 3(C) demonstrates the rheological behavior of  $G'$  and  $G''$  for the hydrogel composite upon increasing the temperature. [Adopted with permission from Fu et al.[112]].



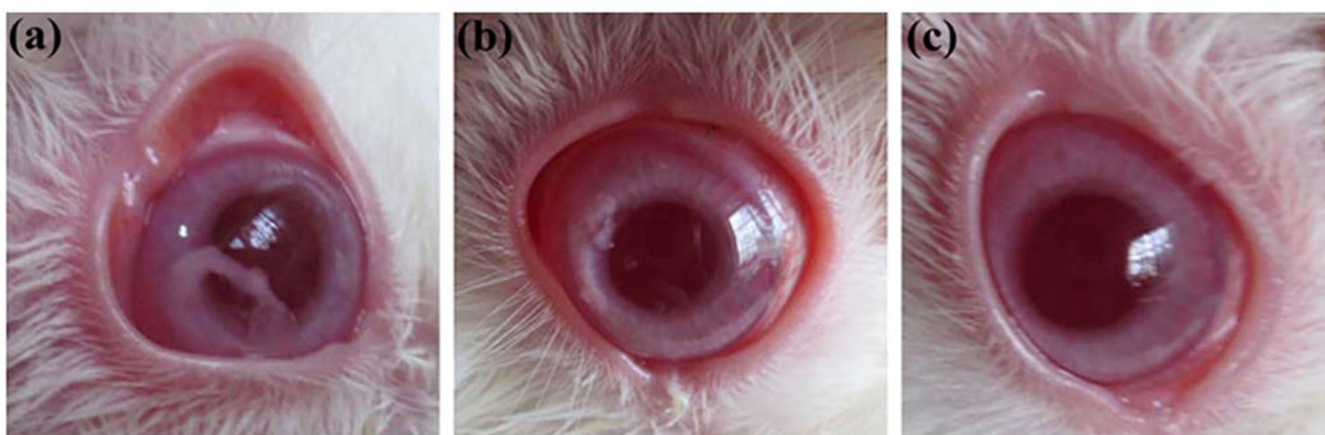


**Fig. 4.** A-K: (A) Schematic illustration of preparation process of TGF- $\beta$ 1/PCEC hydrogel system fabricated with simple procedures; (B) Sol-gel transition of bare PCEC hydrogel. (C) The sol-gel transition of TGF- $\beta$ 1 loaded PCEC hydrogel. (D) *In vitro* injectability test of TGF- $\beta$ 1 loaded PCEC hydrogel; (E) Rheology test of TGF- $\beta$ 1 loaded PCEC hydrogel ( $G'$ : storage modulus,  $G''$ : loss modulus); (F-H) Frontal views of rat knee joint defect samples at 4 weeks, with control, bare PCEC hydrogel and TGF- $\beta$ 1 loaded hydrogel groups arranged from left to right. Defects still caved in, with rough nascent tissue grown in. (I-K) Frontal views of rat knee joint defect samples at 8 weeks, with control, bare PCEC hydrogel and TGF- $\beta$ 1 loaded hydrogel groups arranged from left to right. Cartilage-like tissue filled in defects with gradually obscured boundaries, except for the control group. [Adopted and modified with permission from Zhou et al. [141]]



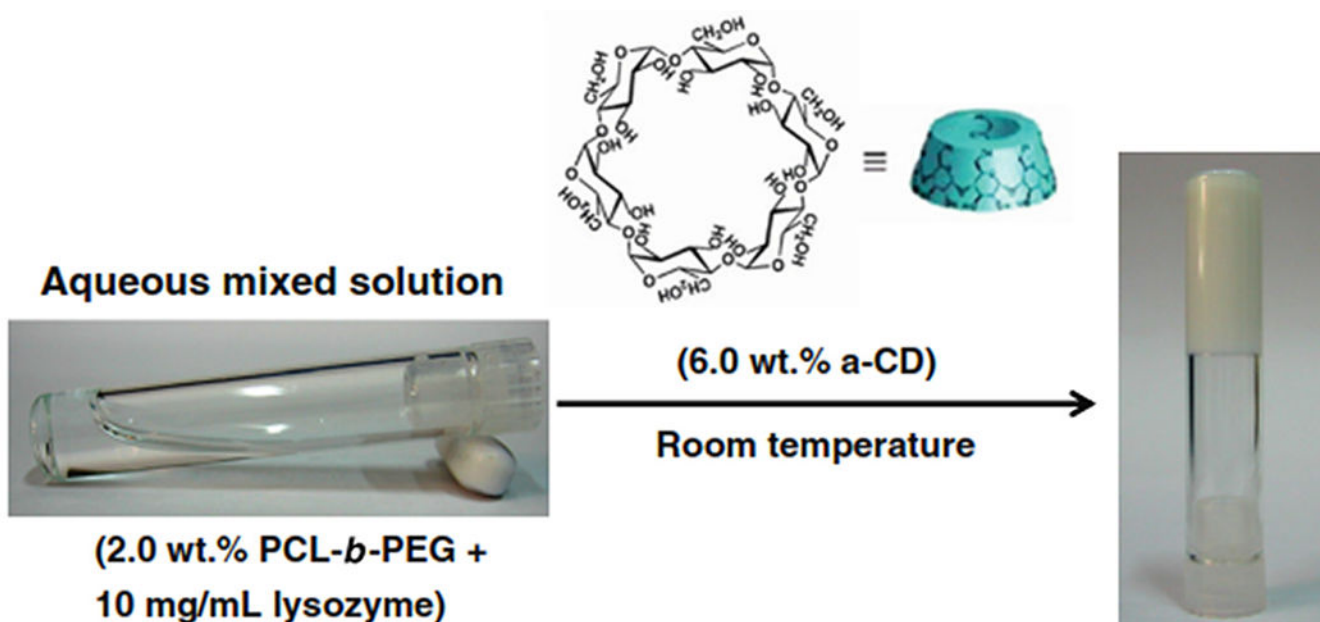
**Fig. 5.** (A) illustration of PTX micelles; (B) (a) normal saline (NS), (b) only micelles (blank), (c) MPEG-PCL/PTX micelles, (d) lyophilized powder of PTX micelles, and (e) re-dissolved freeze-dried PTX micelles. [Adopted and modified with permission from Wu et al.[165]].



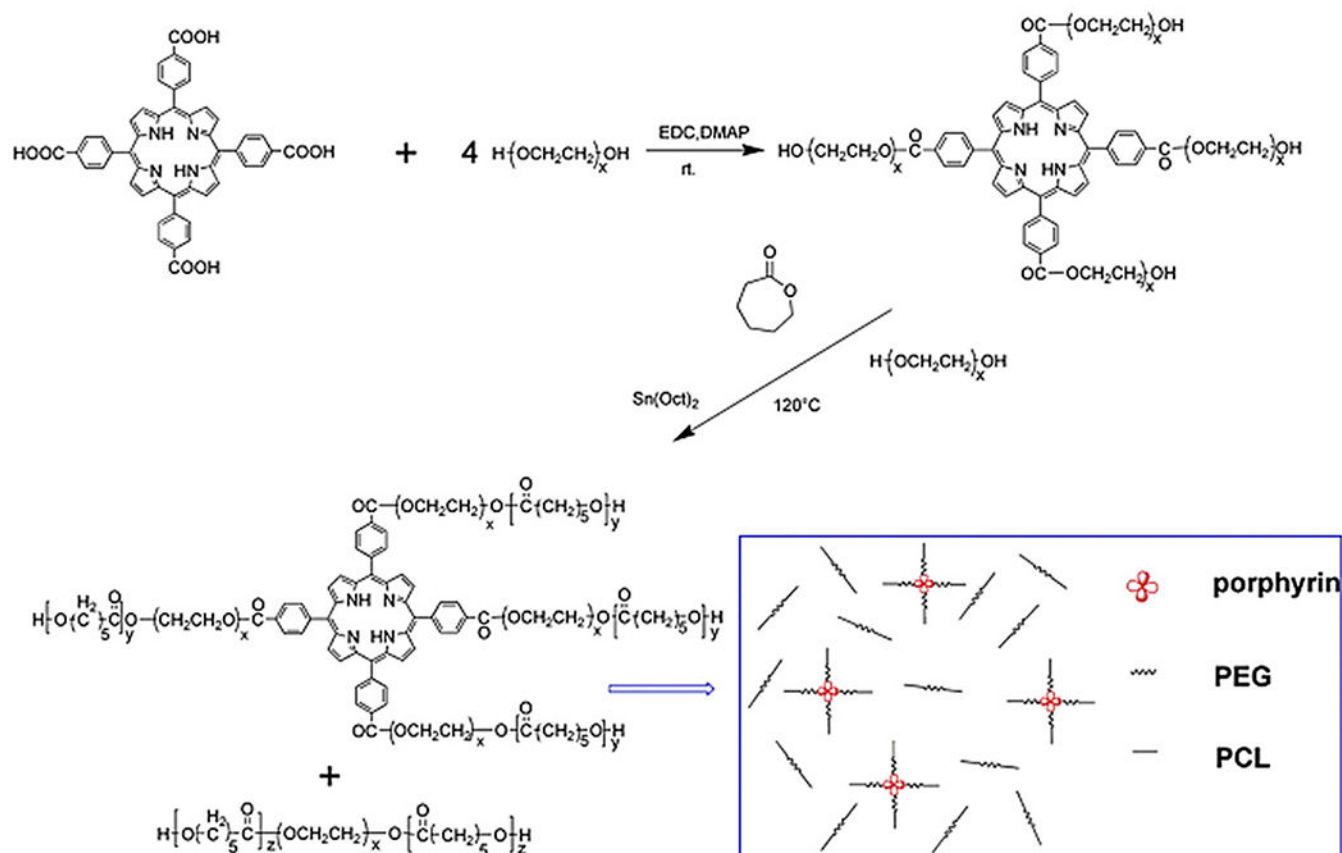


**Fig. 6.**

PECE hydrogel degrades in the anterior chamber. (a) PECE, sol-to-gel transition just after administration into the anterior chamber just above the iris; (b) PECE hydrogel, in 7 days got absorbed in the rabbit eyes; (c) PECE hydrogel completely vanished (absorbed) after 21 days. [Adopted with permission from Han et al. [202]].

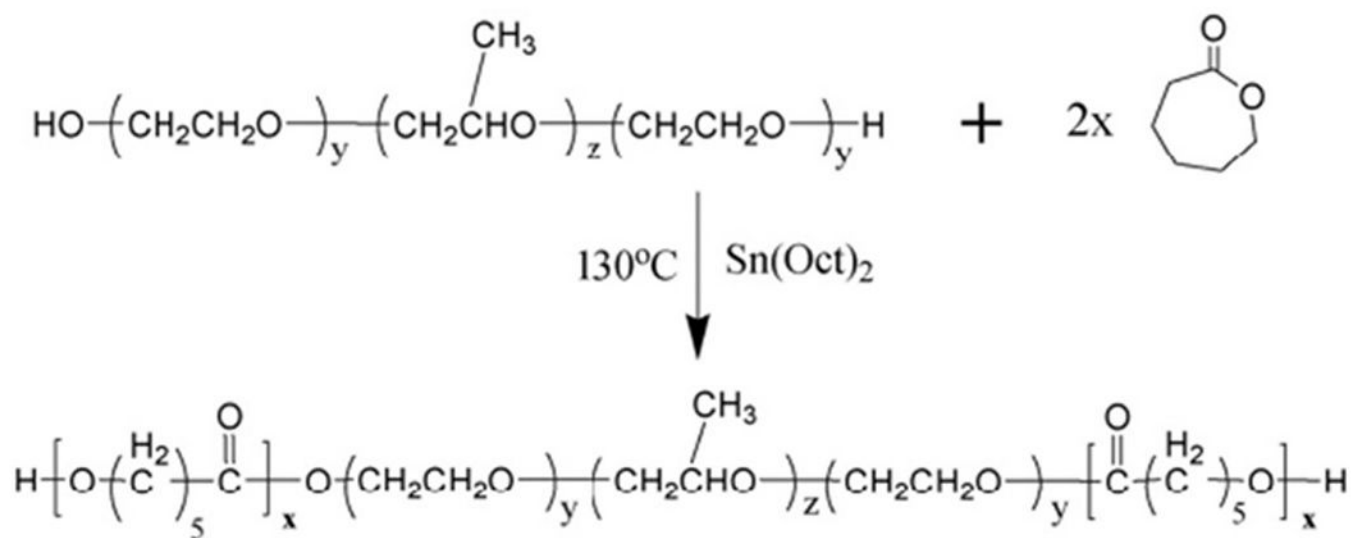


**Fig. 7.**  
Sol-to gel transition of PCL-*b*-PEG/lysozyme after addition of  $\alpha$ -CD. [Adopted with permission from Ma et al.[228]].

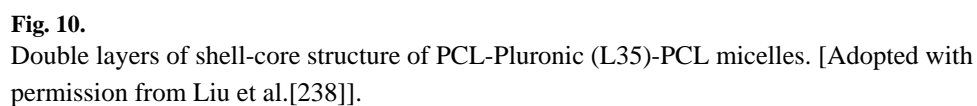


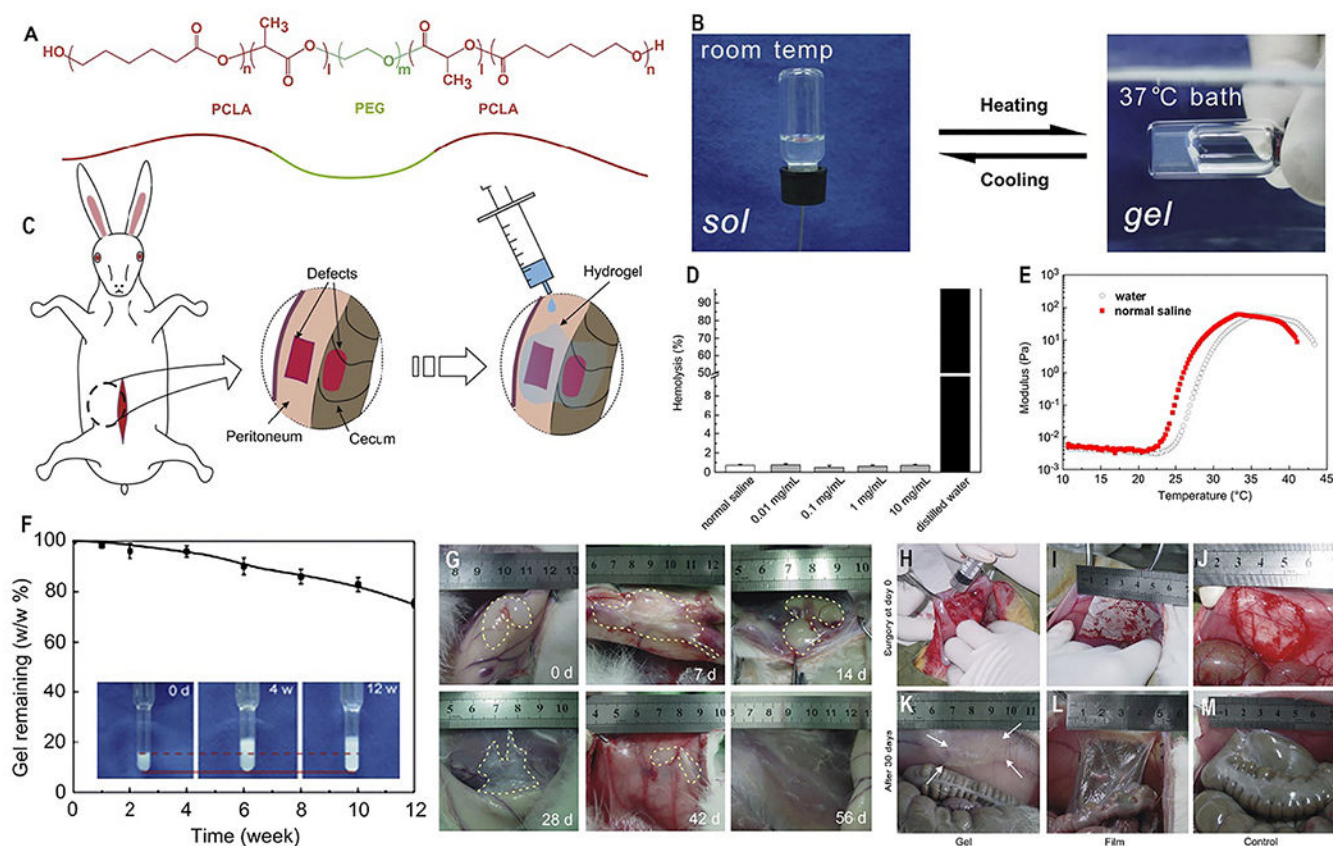
**Fig. 8.**

A synthetic route for the synthesis of a porphyrin-incorporated copolymer. [Adopted with permission from Lv et al.[235]].



**Fig. 9.**  
Schematic representation of the synthesis of PCL-Pluronic (L-35)-PCL copolymers.  
[Adopted with permission from Liu et al.[238]].





**Fig. 11.** (A) Chemical structure of PCLA-PEG-PCLA hydrogel. (B)  $T_{sol-gel}$  behavior, exhibiting sol at room temperature and gel after heating to the body temperature. (C) PCLA-PEG-PCLA hydrogel onto a peritoneal wall defect of a rabbit. (D) Hemolysis study of hydrogel, (E) Storage modulus ( $G'$ ) of PCLA-PEG-PCLA solution. (F) *In vitro* degradation of PCLA-PEG-PCLA hydrogels mass of remaining after 12 weeks. (G) *In vivo* degradation study of hydrogels after subcutaneous injection of 20 wt% hydrogels into the back of the rabbits. The gel remaining indicated by dashed lines. (H) Animal experiments of post-operative adhesions after 30 days, (H & K). Hydrogels applied on the injury sites; (I & L) PLA films applied on the injury sites; (J & M) no barrier materials were used after defects. [Adopted and modified with permission from Zhang et al. [245]].



Table 1

Versatile design strategies to develop PEG-PCL copolymer based thermoresponsive injectable hydrogels as novel drug delivery system.

Type of polymer	Stimulus	Molecular weight	Drug	Sol-gel-sol transition temperature	Drug release behavior	Application	Reference
Triblock copolymer	Temperature	(PCL <sub>1730</sub> -PEG <sub>1500</sub> -PCL <sub>1730</sub> )	–	37 °C	–	Prevention of post-operative adhesion	[245]
Triblock copolymer	Temperature	PEG <sub>960</sub> -PCL <sub>2448</sub> -PEG <sub>960</sub>	5-Fluorouracil	37 °C	Sustained release	Colorectal peritoneal carcinomatosis	[246]
Triblock copolymer	Temperature	PEG <sub>3000</sub> -PCL <sub>2000</sub> -PEG <sub>2000</sub>	Paclitaxel	37 °C	Controlled release	Preventing local breast cancer recurrence and improving incision wound healing in a mouse model	[247]
Tri-block polyols	Temperature	PEG <sub>500</sub> -PCL <sub>2000</sub> -PEG <sub>500</sub>	Polyurethane	–	–	Human tissue engineering applications	[248]
Triblock copolymer	Temperature	PECT	Doxorubicin hydrochloride and Paclitaxel	37 °C	Doxorubicin hydrochloride – initial burst release and Paclitaxel – sustain release	Anti-tumor efficiency	[249]
Triblock copolymer	Temperature	PEG <sub>550</sub> -PCL <sub>2400</sub> -PEG <sub>550</sub>	Honokiol	37 °C	sustain release	Cancer chemotherapy	[70]
Triblock copolymer	Temperature	PCL <sub>1900</sub> -PEG <sub>2000</sub> -PCL <sub>1900</sub> for micelles and PCL <sub>2100</sub> -PEG <sub>1000</sub> -PCL <sub>2100</sub> for hydrogel	Paclitaxel and 5-Fluorouracil	37 °C	Sustained release of PTX and 5-FU, the hydrophilic 5-FU was completely release over a period of 1 week, and the hydrophobic PTX was released slowly with lower cumulative release rate	Colorectal peritoneal carcinomatosis	[250]
Triblock copolymer	Temperature	PCL <sub>1000</sub> -PEG <sub>1000</sub> -PCL <sub>1000</sub>	(TGF)-β1	37 °C	Local sustainable drug delivery system for TGF-β1	<i>In vivo</i> cartilage repair	[251]
Pentablock copolymer	Temperature and pH	PCL <sub>1845</sub> -PEG <sub>2050</sub> -PCL <sub>1845</sub>	Insulin loaded chitosan particle	37 °C	The cumulative release of insulin was around 80% after 30 days, ideal for the treatment of diabetes	Insulin dependent diabetes	[252]
Pentablock copolymer	Temperature	PNIPAAm-PCL-PEG-PCL-PNIPAAm	Cellular culture scaffolds for wound healing	37 °C	Real Time-PCR results shows that 10 and 20% of hydrogel can significantly increase collagen I α1 and collagen III mRNA expression on exposure to hydrogel in fibroblast cells.	Cell scaffold in skin tissue engineering and wound healing	[253]
Diblock copolymer	Temperature	POR-PEG <sub>1000</sub> -PCL	Porphyrin	37 °C	The fluorescence imaging on the treated animal shows irregular fluorescence imaging of hydrogel along with the transition from the liquid state to the solid gel. After the permeation and distribution for 1 day, the hydrogel shrinks and forma a regular gel.	Real time imaging ability conferred to the hydrogels.	[253]

Type of polymer	Stimulus	Molecular weight	Drug	Sol-gel-sol transition temperature	Drug release behavior	Application	Reference
Diblock copolymer	Hydrolytic degradation	PBT-PEG <sub>4000</sub>	–	37 °C	–	Potential application in biomedical field	[254]
Diblock copolymers	Temperature	MPEG-PCL-RGD [Mn = 750 to 2400]	Bone marrow-derived mesenchymal stem cells (BMSCs)	37 °C	Significant increase in the expression levels of collagen I and osteocalcin in BMSCs in the hydrogel.	Bone tissue engineering	[255]
Diblock copolymer	Temperature	PEG-PCL [14:6]	Cultivated articular chondrocytes	37 °C	14:6 weight ratio of PEG:PCL scaffolds were optimal for cartilage tissue formation in terms of collagen type II expression.	Cell growth and tissue regeneration	[14]
Triblock copolymer	Temperature	PEG-PCL-PEG [Mn = 3150]	Intrinsic osteoinductive acellular bone matrix (ABM)	37 °C	Thermosensitive ABM/PECE composites were highly effective towards the repairing of cranial defects with exceptional osteoinductive activity	Bone regeneration	[101]
Triblock copolymer	Temperature	PEG-PCL-PEG	Bevacizumab	37 °C	Sustained release of bevacizumab intracameral injection	Management of postoperative scarring after glaucoma filtration surgery	[256]
Triblock copolymer	Temperature	PEG-PCL-PEG	Collagen and nano hydroxyapatite	37 °C	The hydrogel composite showed excellent capability of guided bone regeneration as compared to the self process, due to the support of collagen scaffold for new bone tissue in-growth and osteoconduction of HA.	Bone regeneration	[112]
Triblock copolymer	Temperature	MPEG-PCL-MPEG	Bone marrow-derived stem cells (BMSCs)	37 °C	The intramyocardial injection of BMSCs loaded hydrogel increased the cell retention and vessel density around the infarct and also prevented the expansion of scar in comparison to the BMSCs injection alone.	Cellular transplantation therapy for myocardial infarction	[257]

It showed Effect of molecular wight of PCL-PEG block copolymers on sol-gel transition and its application.